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# Synthesis, Characterization and Anticorrosion **Studies of Furfuryl Based Methacrylate Copolymer and its Titania Nanocomposites**

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Abstract: Methacrylate based polymer nanocomposite coatings have emerged as a prominent material in corrosion science in recent years. In this article, an attempt has been made to study the anti-corrosive properties of poly(furfuryl methacrylate-coglycidyl methacrylate)/TiO<sub>2</sub> nanocomposite (poly(FMA-co-GMA)/TiO<sub>2</sub>) prepared by in-situ solution polymerization method. To which the oleic acid (OA) modified titanium oxide nanoparticles (OA-TiO<sub>2</sub>) were incorporated during the solution polymerization with different ratios using chlorobenzene as solvent, and AIBN as initiator. The Nuclear magnetic resonance spectroscopy (<sup>1</sup>H and <sup>13</sup>C NMR), Fourier Transform Infrared Spectroscopy (FTIR) and X-ray diffraction (XRD) studies confirmed the formation of poly(FMA-co-GMA) and its TiO<sub>2</sub> composites. The electrochemical characteristics of poly(FMA-co-GMA)/TiO<sub>2</sub> composites film coated on mild steel specimens were investigated and compared with poly(FMA-co-GMA) coated substrate in 3.5% NaCl solution using Tafel and electrochemical impedance spectroscopic (EIS). The morphology of the nanocomposites was confirmed by Scanning electron microscopy (SEM) and Energy dispersive X-ray based elemental mapping. The EIS studies shows that the modified TiO<sub>2</sub> nanoparticles, poly(FMA-co-GMA)/TiO<sub>2</sub> coated on mild steel provides tremendous protection against corrosion in 3.5% NaCl than the copolymer coated (or) bare specimens.

Keywords - Furfuryl methacrylate, Glycidyl methacrylate, Impedance spectroscopy, Tafel plot, Nyquist plot.

# INTRODUCTION

The metallurgical and electrical industries have shown much interest in the topic of corrosion control. Metal corrosion is a pressing problem as it has a significant economic and environmental impact globally. Various forms of corrosion including wet corrosion, crevice corrosion, pitting corrosion, pipeline corrosion, and microbial assisted corrosion were merely associated with maritime environment [1]. To overcome this issue barrier protection characteristic are the prime tool which acts as a layer of protection between the metal and its surroundings. This limits ions from the environment, including water, oxygen, and other ions, get into the metal surface. A prominent area of research in materials science is the development of hybrid materials to enhance coating stability and anticorrosive properties [2].

The most popular technique for providing improved protection for metals from corrosive environments is polymer coating [3]. Developing a high-end engineering polymeric material that is eco-friendly, biosafe, and non-toxic are of utmost importance as protective coatings in corrosion research due to more onerous environmental requirements [4]. This layer serves as a barrier between the metallic surface and corrosive ions in a stringent environment against corrosion. The polymer-based coatings are chromate-free, have great chemical resistance, and have strong adhesion to a variety of metallic substrates makes them suitable options [5].

Methacrylate copolymer coatings are a form of protective coating that improves the corrosion resistance of mild steel, lowers maintenance costs, and offers long-lasting defense against corrosion. Furfuryl methacrylate (FMA) is an important monomer in polymer chemistry and has been studied extensively due to its unique properties and potential applications in numerous fields [6,7]. FMA has an excellent thermal stability and high glass transition temperature, making it suitable for use in high-temperature applications such as adhesives, coatings, and composites.

By reducing porosity and weaving the diffusion channel for corrosive species like oxygen and chloride anions, the incorporation of nanoparticles in polymeric coatings can significantly increase their barrier performance [8, 9]. Composite coatings add a lot of micro-fillers to the polymer matrix in order to attain the desired anti-corrosion qualities [10].

It is obvious that highly dispersed nanocomposites show exceptional anticorrosive behavior, which reflects in this study. The prime theme of this article is to assess the anticorrosive performance of poly(FMA-co-GMA)/TiO<sub>2</sub> composite film coated on mild steel specimens and compared with poly(FMA-co-GMA) coated substrate in 3.5% NaCl solution using Tafel and electrochemical impedance spectroscopic (EIS) studies.

# EXPERIMENTAL Materials

Furfuryl methacrylate purchased from Sigma Aldrich was purified by high vacuum distillation, Glycidyl methacrylate was distilled under vacuum and 2,2'-azobis(isobutyronitrile) (AIBN) (Sigma Aldrich) was purified and recrystallized from ethanol. Chlorobenzene obtained from BDH laboratories, which was used without further purification. The anticorrosive property of the copolymer and its nanocomposites coated on mild steel (MS) in 3.5% (W/V) in NaCl were carried out.

# Synthesis of the Furfuryl methacrylate (FMA)

The synthesis of the Furfuryl methacrylate from furfuryl alcohol and methacryloyl chloride is as follow. The synthesized furfuryl methacrylate is insoluble in water and soluble in most of the organic solvents. The monomer is characterized by FT-IR, <sup>1</sup>H-NMR and <sup>13</sup>C-NMR The structure of the compound is given in **Scheme. 1**.



Scheme. 1: Synthesis of the Furfuryl methacrylate (FMA)

<sup>1</sup>H NMR (300 MHz, CDCl<sub>3</sub>) d <sup>1</sup>/<sub>4</sub> 7.4 (s, 1H, —CH–O– of furan ring), 6.3 (m, 2H, —CH–CH— of furan ring), 4.9 (s, 2H, O– CH<sub>2</sub>– of PFMA), 3.5 (s, 3H, O–CH<sub>3</sub> of PMMA), 0.9–1.9 (different aliphatic protons of the methacrylate unit).

<sup>13</sup>C NMR (300 MHz, CDCl<sub>3</sub>) d <sup>1</sup>/<sub>4</sub> 177.65 (>CO), 149.33 (–C- of furan ring), 110.33 (>CH– of furan ring), 143.60 (>CH– of furan ring), 125.90 (O–CH<sub>2</sub> of PFMA unit), 52.12 (O–CH<sub>3</sub> of PMMA unit), 58.68 (–CH<sub>2</sub>– of the methacrylate unit), 45.15 (>C< of the methacrylate unit), 18.91 and 17.85 (atactic structure of a-CH<sub>3</sub> side chain).

FT-IR (KBr): 3 122, 2 957, 1 727, 1 501, 1 153, 1 012, 919, 816, 746 cm<sup>-1</sup>.

# Surface modification of Titanium oxide nanoparticles

Surface modification of TiO<sub>2</sub> nanoparticle were carried out with oleic acid [11]. 50ml of 2% solution of oleic acid was prepared in ethanol. To this solution 0.5g of Titanium oxide nanoparticles (TiO<sub>2</sub> NPs) was added and stirred at 50<sup>o</sup>C. After 4 h the contents were centrifuges (1 x  $10^{6}$ rpm) the grafted TiO<sub>2</sub> nanoparticle were collected and washed with ethanol followed by acetone (5 x 30ml) and dried under reduced pressure (Scheme. 2).

## Synthesis of Poly (FMA-co-GMA):

The copolymer was synthesized by free radical solution polymerization [12, 13]. The required amount of monomers Furfuryl methacrylate and Glycidyl methacrylate were dissolved in chlorobenzene along with the initiator AIBN in a polymer tube as shown in **Scheme 3**.

# Synthesis of Furfuryl methacrylate nanocomposite:

Poly(FMA-co-GMA)/OA-TiO<sub>2</sub> were prepared (**Scheme. 4**) by the addition of different moles of Furfuryl methacrylate, Glycidyl methacrylate (0.20:0.80, 0.50:0.50, 0.80:0.20) and 5 wt.% of OA-TiO<sub>2</sub> in 50ml chlorobenzene solvent. Further the solution was kept for ultra-sonication for 30 min, then polymerization was carried out at 60°C using AIBN as radical initiator under N<sub>2</sub> atmosphere (2hr). After cooling, white coloured amorphous powder was obtained by the addition of hexane, it was further re-precipitated from chloroform to get the pure product.



## Scheme. 2: Surface Modification of TiO<sub>2</sub>

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Scheme. 3: Synthesis of the poly(FMA-co-GMA)



Scheme. 4: Synthesis of the poly(FMA-co-GMA)/TiO<sub>2</sub> Nanocomposite

# **RESULT AND DISCUSSION:** <sup>1</sup>H NMR Spectroscopy:

Synthesized poly(FMA-co-GMA) were characterized by <sup>1</sup>H NMR spectra taken on a Bruker-Instrument-NMR Spectrometer (DPX400) with DMSO as the solvent. In **Fig 1:** 



**Fig. 1:** <sup>1</sup>H-NMR Spectrum of the poly(FMA-co-GMA)

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<sup>1</sup>H NMR (300 MHz, CDCl3) d <sup>1</sup>/<sub>4</sub> 7.4 (s, 1H, —CH–O– of furan ring), 6.3 (m, 2H, —CH–CH— of furan ring), 4.9 (s, 2H, O– CH<sub>2</sub>– of PFMA), 3.5 (s, 3H, O–CH<sub>3</sub> of PMMA), 0.8–1.9 (different aliphatic protons of the methacrylate unit).

<sup>1</sup>H NMR spectrum of poly(FMA-co-GMA) confirms the presence of pyrrole ring (m, 4H), The peaks at 0.7 to 1.9 ppm are due to the different saturated protons ( $-CH_3$  and  $-CH_2-$ ) of the methacrylate units of poly(FMA-co-GMA). The peaks at 7.4 and 6.3 ppm correspond to the different aromatic protons of furan ring. The spectrum shows two signals at 4.28 and 3.79 ppm, which are due to  $-COOCH_2$  – group at GMA units [14]. The peak at 3.209 ppm is due to the methylene proton of the epoxy group. The methylene protons of the epoxy group show signals at 2.72 and 2.60 ppm. The backbone methylene groups show signals at 1.54 – 2.51 ppm. All of the copolymers were soluble in many of the organic solvents, such as THF, CHCl<sub>3</sub> and Toluene.



Fig. 2: <sup>13</sup>C-NMR Spectrum of the poly(FMA-co-GMA)

 $^{13}$ C NMR (300 MHz, CDCl<sub>3</sub>) d <sup>1</sup>/<sub>4</sub> 177.30 (>CO), 110.53 (>CH– of furan ring), 143.30 (>CH– of furan ring), 125.90 (O–CH<sub>2</sub> of PFMA unit), 44.73 (>C< of the methacrylate unit), 18.91 and 17.85 (atactic structure of a-CH<sub>3</sub> side chain).

The <sup>13</sup>C NMR spectrum of the poly(FMA-co-GMA) is shown in Figure 2. The ester carbonyl of GMA appeared at 177.7 ppm. The aromatic carbons of FMA unit in copolymer appeared at 143.3, 110.53, respectively. The methylene group of the epoxy group in GMA unit shows signal at 66.3 ppm.

# **FT-IR Spectroscopy:**

A transmission vibrational spectrum of OA-TiO<sub>2</sub>, poly(FMA-co-GMA), and poly(FMA-co-GMA)/TiO<sub>2</sub> are shown in Fig. 3.



Fig. 3: FT-IR Spectrum of the a) OA-TiO<sub>2</sub>, b) poly(FMA-co-GMA), c) poly(FMA-co-GMA)/TiO<sub>2</sub>

FTIR analysis was employed to study the characterization of OA-TiO<sub>2</sub> nanoparticles, is shown in Figure 3a. The first band is the broadest, and is observed at 3500 cm<sup>-1</sup>, corresponding to the stretching vibration of the hydroxyl group O-H of the TiO<sub>2</sub> NPs. The second band is observed around 1630 cm<sup>-1</sup>, corresponding to bending modes of water Ti-OH and Ti-O bending mode vibration observed at 483cm<sup>-1</sup>; the last is a prominent peak at 1398 cm<sup>-1</sup> related to Ti-O modes [15]. The long alkyl groups in oleic acid were characterized in the peaks of 2929 cm<sup>-1</sup> and 2857 cm<sup>-1</sup>. Moreover, the adsorption peaks in 1713 cm<sup>-1</sup> attributed to carbonyl groups of oleic acid in OA-TiO<sub>2</sub> nanoparticles [15] and -C=C- peak appeared at 1610cm<sup>-1</sup>.

In Fig 3b. Shows FTIR Spectrum of poly(FMA-co-GMA). The vibration bands observed at 2996, 2937, and 1258 cm<sup>-1</sup> are for the  $-CH_3$  asymmetric and symmetric stretching and symmetric bending vibrations of methacrylate units, respectively. The vibration peak at 908 cm<sup>-1</sup> is assigned to -CH bending vibrations, and the peak at 556 cm<sup>-1</sup> is assumed to be related to the CH<sub>3</sub> (C–O) group. The C–H wagging vibration modes are confirmed by the presence of a band at 750 cm<sup>-1</sup> [16]. Carbonyl groups gave strong peaks at 1723, 1144, and 1258 cm<sup>-1</sup>, attributed to the stretching of the glycidyl methacrylate ester group. In Fig 3c, shows respective peaks of polymer nanocomposites confirms the FTIR Spectrum of poly(FMA-co-GMA)/TiO<sub>2</sub>. **XRD Studies** 

XRD pattern of the crystalline structure of OA-TiO<sub>2</sub> nanoparticle, poly(FMA-co-GMA) and poly(FMA-co-GMA)/TiO<sub>2</sub> nanocomposites were shown in fig 4. the OA-TiO<sub>2</sub> exhibits the prominent peaks at around 24.61, 27.61, 36.26, 41.41, 54.56, 56.86 and 69.21  $\theta$  [11]. The average grain sizes (D) were determined by the Debye-Scherrer's equation.

 $Dc = 0.9\lambda/\beta cos\theta$ 

Where  $\lambda$  is the X-ray wavelength and is the  $\beta$  width of diffraction peak at its half maximum intensity.

In Figure 4b, the coexistence of a broad amorphous peak from poly(FMA-co-GMA) and suppressed crystalline peaks from  $TiO_2$  confirms the excellent distribution and compatibility of  $TiO_2$  within the resulting nanocomposite. The diffractogram of OA- $TiO_2$  nanoparticles and poly(FMA-co-GMA)/ $TiO_2$  composites shows characteristic peaks of the anatase phase (major peaks: 25.4°, 38.0°, 48.0°, and 54.7°) and the rutile phase (2 $\theta$  at 27.4°, 36.1°, 41.3°, and 54.4°) which demonstrating that the incorporation of  $TiO_2$  nanoparticles in the poly(FMA-co-GMA)/ $TiO_2$  composites is mixed types of anatase/rutile  $TiO_2$  composites. The size of the crystals was 17nm based on estimation using Scherrer equation.



**Fig. 4:** XRD Spectrum of the a) OA-TiO<sub>2</sub>, b) poly(FMA-co-GMA), c) poly(FMA-co-GMA)/TiO<sub>2</sub> **Thermal Studies:** 

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The thermal stability of the different copolymers and its nanocomposites were investigated by TGA. Figure 5 shows the TGA curves of poly(FMA-co-GMA) and poly(FMA-co-GMA)/TiO<sub>2</sub> the obtained results were summarized in Table 1. The TGA curve clearly shows that all the thermograms followed the three or four stage decomposition patterns.

It can be seen that the increase of FMA content increases the initial decomposition temperature, also increases the residual amount of char left out. The ratios of 20:80, 0.50:0.50 (FMA/GMA) copolymers showed initial weight loss, 3.89%, 3.75% at 101.5°C, 105.3°C respectively due to evaporation of water or solvent. The second stage decomposition starts from 302°C to 320°C, it's due to cleavage of furfuryl pentane groups, and the third stage decompositions start from 388 to 430°C due to cleavage of polymer backbone chain.

# Table 1. TGA Data of Copolymer (FMA-co-GMA) and its nanocomposites.

Obviously, it is observed that the thermal decomposition temperatures of polymer (0.80:0.20 (FMA/GMA)) are higher than that of poly(FMA-co-GMA)/TiO<sub>2</sub> composite material or else simply thermal stability of the copolymer increases as increases of furfuryl moieties in the copolymers.



S.No.	poly(FMA-co-	Tg	IDT	Temperature ( <sup>0</sup> C) at weight loss (%)				
	GMA)	( <sup>0</sup> C)	( <sup>0</sup> C)	10%	30%	50%	70%	90%
1.	poly(FMA-co- GMA)-20:80	329	169	244	304	349	389	489
2.	poly(FMA-co- GMA)-50:50	327	175	265	305	340	394	555
3.	poly(FMA-co- GMA)-80:20	322	183	293	363	408	448	673
4.	poly(FMA-co- GMA)-50:50/TiO2	306	182	282	332	387	424	657

Fig. 5: TGA Spectrum of the a) OA-TiO<sub>2</sub>, b) poly(FMA-co-GMA), c) poly(FMA-co-GMA)/TiO<sub>2</sub>

Fourier Emission- Scanning Electron Microscope (FE-SEM):



Fig. 6: SEM Images of solid a) poly(FMA-co-GMA), b) poly(FMA-co-GMA)/TiO<sub>2</sub>, c) OA-TiO<sub>2</sub>,

FE-SEM indicates that the surface of poly(FMA-co-GMA) exhibits a smooth and homogeneous texture (Fig 6a), while the poly(FMA-co-GMA)/TiO<sub>2</sub> (Fig 6b) composites display a highly rough and heterogeneous surface, indicating the successful incorporation of surface-modified nanoparticles.

The observed significant morphological changes in the composites can be attributed to the strong influence of the hydrophobic nature of  $TiO_2$  nanoparticles added to the Poly(FMA-co-GMA) system, which enhances the compatibility between the polymer and OA-TiO<sub>2</sub> and impacts the overall composite morphology [11]. When examining the SEM image, the nanoparticles are evenly incorporated through entire surface of polymer composites can be visually observed at high magnification.

The FESEM images and EDX analysis of the surfaced modified OA-TiO<sub>2</sub> NPs show in Figure 6c. which confirms that the OA-TiO<sub>2</sub> NPs are spherical and highly uniform in size. In addition, the EDX analysis confirms, the peak of C shows the existence of OA functionality on the surface of TiO<sub>2</sub> NPs. Furthermore, the EDX analysis indicates that titanium (Ti), oxygen (O), and carbon (C) are the major constituents of the nanoparticles. which is also confirmed by FTIR.



Fig. 7: SEM images of mild steel samples: (a) polishing surface (b) after corrosion studies in 3.5% NaCl solution, (c, d, e) polymer coated Mild steel before & after corrosion studies and its EDAX, (f) polymer coated Mild steel before corrosion studies

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Surface morphology of bare and coated mild steel were studied by scanning electron microscopy before and after the corrosion studies in 3.5% NaCl. [17] Figure 7(a) displays the SEM image of the polished Mild steel sample prior to exposure to the corrosive environment. The image reveals a smooth surface with some abrasion scratches present. In contrast, Figure 3(b) exhibits significant damage on the surface of the mild steel sample resulting from the formation of corrosion products after immersion in a 3.5% NaCl solution, which illustrates the aggressive attack of the corrosive medium on the mild steel surface.

In Fig. 7(a and b) shows the SEM micrograph of  $poly(FMA-co-GMA)/TiO_2$  composite were coated before and after the corrosion. The surface of any appears smooth without the formation of significant pits, indicating the protective nature of the composite coating. On the other hand, Figure 7(c) illustrates the surface topography of the poly(FMA-co-GMA) polymer coating before and after corrosion testing. The surface exhibits roughness with numerous deep pits, which likely migrated from the substrate to the coating surface during the corrosion process through micro-pores. Upon closer inspection at higher magnification, the pits display distinct corrosion product morphology, resembling flower-like microstructures, accompanied by smaller particles that appear white in color.

The outcome of the experiment showed an improvement in the surface coverage on the mild steel, resulting in reduced contact between the steel and the corrosive medium. Thus, the presence of a protective layer with good absorbent properties effectively inhibited the corrosion of the mild steel [18]. However, the nanocomposite coating exhibited a smooth and uniform surface without any visible pits after the corrosion studies, indicating a significant enhancement in the overall corrosion resistance of the coating.

In addition, Elemental mapping of  $poly(FMA-co-GMA)/TiO_2$  composite coating surface are shown in Figure 8. The red dots, green dots, and pink dots corresponding to the distribution states of Carbon (C), Oxygen (O), and Titania (Ti) indicated that prepared composites TiO<sub>2</sub> nanoparticles are successfully adsorbed on (poly(FMA-co-GMA) surface.

The EDAX- line maps provide further evidence that, the  $TiO_2$  nanoparticles were evenly distributed across the coated surface and homogeneously dispersed within the polymer matrix, as depicted in Fig 9. [19].



Fig. 8.  $poly(FMA-co-GMA)/TiO_2$  composite coating on Mild steel.



Fig. 9. EDAX-line analysis of poly(FMA-co-GMA)/TiO2 composite coating on Mild steel.

# **Electrochemical Characterization:**

# **Electrochemical Impedance Spectroscopy**

The Nyquist plots of poly(FMA-co-GMA) and poly(FMA-co-GMA)/TiO<sub>2</sub> composite coated on mild steel specimens with different feed ratios in 3.5 wt.% NaCl solution medium is presented in Fig. 10. The EIS measurements were carried out and compared with the uncoated sample, which showed better anticorrosion efficacy. The impedance spectra of the coated specimens exhibited notable differences in both shape and size compared to the uncoated specimen. Table 2 shows the prominent existence of a substantial capacitive loop, which can be ascribed to the presence of the polymeric coating applied on the metal surface. By increasing the concentration of FMA, the capacitive loop diameter at high frequency also increases, which is related to high corrosion resistance [20]. Moreover, the sample coated with poly(FMA-co-GMA)/TiO<sub>2</sub> composite shows the largest loop compared with other polymer coated samples which is coincidence with inhibition efficiency (IE%).

Interestingly, increasing the concentration of FMA in the copolymer results in a larger capacitive loop diameter at high frequencies, indicating enhanced corrosion resistance. Additionally, the sample coated with poly(FMA-co-GMA)/TiO<sub>2</sub> composite exhibits the largest loop compared to other polymer-coated samples, coinciding with a higher inhibition efficiency (IE%).

This enhancement can be attributed to the incorporation of  $TiO_2$  nanoparticles throughout the entire surface of the copolymer film significantly enhances the charge transfer resistance, preventing the penetration of the electrolyte [21].



Fig. 10. Nyquist plot of poly(FMA-co-GMA) and its TiO<sub>2</sub> composite coating on Mild steel.

The phase angle plot of the bare substrate, shows one time constant, suggesting a severe corrosion reaction. After the substrate is coated with poly(FMA-co-GMA) coatings, one time constant appears at high-frequency region. Typically, the time constant in the high frequency region reflects the barrier performance of the coating, while that in the low frequency region is indicative of the corrosion rate. Compared with the polymer coating [22], the time constant of the TiO<sub>2</sub> composites coating shifts positively from  $10^3$ Hz in the high frequency region, suggesting the improved barrier performance and lower corrosion rate.

To comprehend the experimental impedance response of each coated surface during the testing period, the impedance data was fitted to equivalent electrical circuits (Fig 11). These circuits were designed to consider either one or two time constants, allowing for a detailed analysis of the passive electrical elements involved. The proposed circuits proved to be suitable for explaining the behavior of the passive metal/electrolyte interface after prolonged immersion in a 3.5% NaCl solution [23]. The electrical components used to fit the impedance data in the equivalent circuits, are  $R_e$ , which is the electrolyte resistance, and CP and  $R_p$ , which are the constant phase element and the resistance linked with the time constant associated with the metallic surface/electrolyte electrochemical interface.

The Nyquist plots show a depressed semi-circle with the center under the real axis, whose size increases with different mole ratios of copolymer coatings, indicating a charge transfer ( $R_{ct}$ ) process mainly controlling the corrosion of the mild steel. The electrochemical parameters, charge transfer resistances ( $R_{ct}$ ) and double layer capacitances ( $C_{dl}$ ) were calculated from the Nyquist plots [18]. The corrosion inhibition efficiency (IE%) was calculated using following equation. An increase in the concentration of polymer increases the corrosion inhibition efficiency, but the slope of the impedance curve pattern does not show any change.

S.NO	Polymer/Composite	$R_s(\Omega/cm^2)$	$R_c(\Omega/cm^2)$	$C_{dl}(F \text{ cm}^{-1})$	Inhibition Efficiency (%)
1.	Uncoated Mild steel	50.94	130.8	7.583e-7	-
2.	0.20:0.80 - poly(FMA-co-GMA)	68.46	1418	5.24e-7	90.78
3.	0.50:0.50 – poly(FMA-co-GMA)	69.25	1801	6.33e-7	92.73
4.	0.80:0.20 - poly(FMA-co-GMA)	68.43	1902	1.13e-7	93.12
5.	0.50:0.50 - poly(FMA-co-GMA)/TiO <sub>2</sub>	77.65	7756	1.662e-7	98.31

Table: 2. Impedance parameters of poly(FMA-co-GMA) and its TiO<sub>2</sub> composite coating on Mild steel in 3.5% NaCl.



**Fig. 11.** Bode plot of poly(FMA-co-GMA) and its TiO<sub>2</sub> composite coating on Mild steel.  $IE(\%) = \frac{(R_{ct}-R_{ct})}{R_{ct}}X 100$ 

# **Potentiodynamic Polarization studies**

The potentiodynamic polarization studies were carried out for MS specimens, both for the polymer coated and uncoated of different mole ratios (0.20:0.50, 0.50:0.50 and 0.80:0.20) of poly(FMA-co-GMA) and their corresponding poly(FMA-co-GMA)/TiO<sub>2</sub> composites in 3.5% NaCl solution (Fig 12). The electrochemical kinetic parameters such as corrosion current density (*Icorr*), corrosion potential (*Ecorr*), Inhibition efficiency ( $\eta$ %), and Corrosion (*CR*) were calculated and were given in Table 3. For the uncoated Mild steel, the *Ecorr* and *Icorr* value are 0.527 and 3.462e-5 respectively and substantially decreased by coating of polymers [24]. It shows that polymers reduce the corrosion current densities and consequently exhibits the corrosion inhibition property. The corrosion inhibition efficiency ( $\eta$ %) was calculated from the corrosion current density values using the following eqn.

$$\eta(\%) = \frac{(I'_{corr} - I_{corr})}{I'_{corr}} X \ 100$$

where *I'corr* and *Icorr* are the corrosion current densities of uncoated and coated mild steel, respectively.

Corrosion current densities of poly(FMA-co-GMA)/TiO<sub>2</sub> composites are lower than those of the pristine polymers (0.20:0.80, 0.50:0.50 and 0.80:0.20). This clearly indicates that the TiO<sub>2</sub> nanoparticles embedded poly(FMA-co-GMA) adsorbs more on the metal surface, thus reducing the metal corrosion and reduce the active metal surface available for the attack of corrosive medium as well as metal dissolution.

It is due to this fact that the  $TiO_2$  nanoparticles are arranged uniformly and cover more metal surface than the pristine polymer. It is worth mentioning that the poly(FMA-co-GMA)/TiO<sub>2</sub> composites showed an excellent inhibition activity of 99.96% whereas 0.20:0.80, 0.50:0.50 and 0.80:0.20 ratios of poly(FMA-co-GMA) polymeric coating showed 74.04%, 94.80% and 98.48%. As the feeding ratios of FMA are increased in polymer, the corrosion potential (Ecorr) and corrosion current density (Icorr) decreased [24]. Hence, the corrosion inhibition increases, and surface coverage also increases with FMA concentration.



Fig. 12. Tafel plot of poly(FMA-co-GMA) and its TiO<sub>2</sub> composite coating on Mild steel.

Table: 3. Potentiodynamic polarizations param	eters of poly(FMA-co-GM	(A) and its $TiO_2$ co	omposite coating on Mild steel in
	3.5% NaCl.		

Polymer/Composite	E Corr (A)	Icorr (A)	Inhibition Efficiency (%)	Corrosion rate(mil/yr)
Uncoated Mild steel	0.527	3.462 x 10 <sup>-5</sup>	-	$2.22 \times 10^3$
0.20:0.80 - poly(FMA-co-GMA)	0.462	8.986 x 10 <sup>-6</sup>	74.04	5.767 x 10 <sup>2</sup>
0.50:0.50 - poly(FMA-co-GMA)	0.449	1.800 x 10 <sup>-6</sup>	94.80	$1.55 \ge 10^2$
0.80:0.20 - poly(FMA-co-GMA)	0.433	5.273 x 10 <sup>-7</sup>	98.48	3.384 x 10 <sup>1</sup>
0.50:0.50 – poly(FMA-co- GMA)/TiO <sub>2</sub>	0.365	1.250 x 10 <sup>-5</sup>	99.96	8.0220

## **CONCLUSION:**

In this manuscript, we have successfully synthesized poly(FMA-co-GMA) and poly(FMA-co-GMA)/TiO<sub>2</sub> composites [0.20:0.80, 0.50:0.50 and 0.80:20] by free radical solution polymerization and *in-situ* polymerization techniques. Spectral and morphological studies like FT-IR, <sup>1</sup>H- and <sup>13</sup>C-Nuclear magnetic resonance spectroscopy, XRD and FE-SEM analysis. TGA studies indicated that the 0.80:0.20 ratio was more thermally stable than other copolymer compositions due to increases of Furfuryl ring in the copolymer system. The corrosion resistance performance of the Poly(FMA-co-GMA) and its TiO<sub>2</sub> nanocomposites coating in 3.5% NaCl solution was studied, which showed that nanocomposite has provided an effective way to fabricate a good anticorrosive coating in aggressive conditions.

The anti-corrosion study of the poly(FMA-co-GMA)/TiO<sub>2</sub> composite film coated on mild steel specimens were investigated and compared with poly(FMA-co-GMA) coated substrate in 3.5% NaCl solution using Tafel and electrochemical impedance spectroscopic (EIS) studies. Surface morphology studies reveals that, the nanocomposite coating exhibited a smooth and uniform surface without any visible pits after the corrosion studies, indicating a significant enhancement in the overall corrosion resistance of the coating.

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