

STUDY ON EFFECT OF VARIATION OF DIISOCYANATE AND NCO/OH MOLAR RATIO ON MECHANICAL, THERMAL AND ELECTRICAL PROPERTIES OF INTERPENETRATING POLYMER NETWORKS DERIVED FROM BIOMASS

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ABSTRACT

Polymers derived from biomass are now a matter of the utmost importance from both commercial and academic points of view. Interpenetrating polymer networks were prepared from Castor oil, toluene 2,4-diisocyanate, Acrylonitrile with such polymers were investigated by FTIR spectroscopy. All these IPNs films were characterised by resistance to chemical reagents, swelling behavior, mechanical studies such as tensile strength, Young's modulus, elongation at break (%) and soil burial. Thermal stability behavior of the polymer degradation was studied by thermogravimetric analysis (TGA), and differential thermal analysis (DTA). Dielectric properties such as dielectric constant, dielectric loss and dielectric loss tangent were also estimated. The scanning electron microscopy (SEM) micrographs revealed changes in the surface morphology of the polymers.

Key words: Castor oil; Acrylonitrile; cross linked polyurethanes; interpenetrating polymer networks; crosslinker N,N'-methylenebis acrylamide;

INTRODUCTION

Interpenetrating polymer networks (IPNs) constitute a rapidly developing branch of polymer blend technology which may intimately combine the properties of two crosslinked polymers in a network form. An IPN structure is obtained when at least one polymer network is synthesized and/or crosslinked in the immediate presence of the other. Proper combination and interpenetration of polymer networks will enable to enhance certain properties which are otherwise difficult to achieve by chemical variations in the polymer chain. The IPNs synthesized by the combination of various chemical types of polymer network exhibit varying degree of phase separation depending on the compatibility of component polymers [2-4]. In

recent time IPNs synthesized from castor oil have been paid good attention in the industrial applications (Begum and Siddaramaiah, 2004). IPNs have been used as damping materials⁶⁻⁸, biomedical materials⁹, gas transport membranes¹⁰. Polyurethanes (PU) comprise a class of materials which can vary widely from rubbery to glassy thermoplastics and from linear polymer to being a thermoset. By varying the processing conditions the properties of polyurethanes can be tailor-made to produce desirable molecular structure and physical properties¹¹⁻¹³. Castor oil and its derivatives have been used in the preparation of rigid, semi-rigid and flexible polyurethane foams¹⁴. Castor oil polyurethanes have been used for the synthesis of semi IPNs and sequential IPNs¹⁵.

MATERIALS AND METHODS

Castor oil, toluene 2,4-diisocyanate (TDI), N,N-methyleneacrylamide and benzoylperoxide (sigma aldrich) were used. Acrylonitrile (AN) monomer from HPLC chemicals. All other reagents were of analytical grade.

.Preparation of Polyurethane prepolymer (PPU)

A reaction kettle, under dry nitrogen, was charged with TDI and castor oil of varying ratio of NCO/OH was added slowly with stirring. The reaction was carried out at 45°C for 1h.

Synthesis of Interpenetrating Polymer Networks (IPNs)

The PU and AN in different weight to weight ratios were separately taken in presence of the initiator and the crosslinker. The mixture was constantly stirred at room temperature for 15min. Thereafter, the temperature was increased to 75°C with stirring for about 1 h which was poured into a glass mould and kept in an oven at 75°C for 24 h. The thin film was formed, cooled and removed from the mould.

Table 1 Data on feed composition of individual IPNs

Sample code	NCO/OH	Content of polyurethane (wt%)	Content of monomer (wt%)
IPN-1CA	1.6	25	75
IPN-2CA	1.6	35	65
IPN-3CA	1.6	45	55
IPN-4CA	1.8	25	75
IPN-5CA	1.8	35	65
IPN-6CA	1.8	45	55
IPN-7CA	2.0	25	75
IPN-8CA	2.0	35	65
IPN-9CA	2.0	45	55

Characterization Methods

Infrared spectra

Infrared spectra of the synthesized IPNs in KBr pellets were obtained from Shimadzu FTIR 4200 series spectrophotometer.

Mechanical properties

The tensile strength, Young's modulus and elongation at break were measured at room temperature using Instron Hounsfield universal testing machine model 4204 as per ASTM D-638 method.

Resistance to chemical reagents

Acid, alkali and solvent resistance were estimated according to ASTM-D543-67 method. Samples were hung in the reagent for seven days and tested for change in weight and for their film appearance¹⁶.

Determination of Stability in Organic Solvents

The stability of the newly prepared biopolymers in common organic solvents acetone, toluene, methyl ethyl ketone and dimethyl acetamide (DMA) were carried out according to ASTM C 267. The conditioned samples were kept in 100 ml of the organic solvent for the period of 60 days. The medium was changed once in a week. The weight loss was determined¹⁷.

Thermogravimetric analysis

In the present work, TGA/DTA studies were carried out using SDT Q 600V 8.3 build 101 instrument at the heating rate of 10 C/min from ambient to 700 C in nitrogen atmosphere using 2-5 mg of the sample¹⁸.

Electrical properties

The IPNs prepared from castor oil and toluene-2,4-diisocyanate reacted with acrylic monomer like methyl methacrylate were taken for experiment to determine electrical properties. Dielectric properties such as dielectric constant and loss factor were evaluated by taking thin film of 8 mm diameter (d) and thickness (t) of 0.3 mm, using Hioki-3532-50 LCR HiTester Instrument.

Soil burial test

The replicate pieces of the sample (5 cm x 3 cm) were buried in the garden soil at the depth of 30 cm from the ground surface for 3 months, inoculated with the sewage sludge having ability to adhere and degrade the polymer film. The test specimen was periodically removed from the soil and the specimen was then gently washed to remove attached soil and dust after being dried in vacuum oven¹⁹. The extent of degradation was examined by weight loss and surface observation. Scanning Electron Microscope (SEM) was used for assessing surface damages of polymeric sheet subjected to soil burial test.

Scanning electron microscopy

To produce the SEM image, the electron beam is swept across the area being inspected, producing many such signals. These signals are then amplified, analyzed, and translated into images of the topography being inspected. Finally, the image is shown on a CRT20.

The surface morphology of the samples were done by a Joel scanning electron microscope of model JSM-6390LV SEM after platinum coating on the surface.

RESULTS AND DISCUSSION

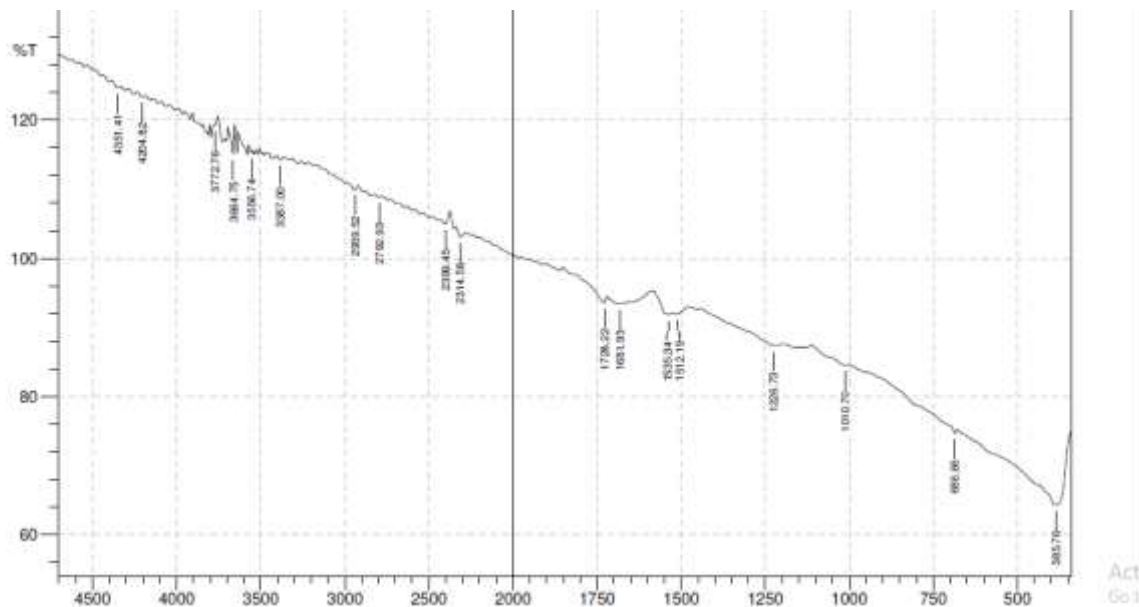
FT-IR Spectral Analysis

FTIR provides a method for identifying the chemical bonds and units present in the chemical structure. This method can be used to identify whether any new bonds or units have been constructed during the curing of the mixtures. In particular, the goal is to identify whether there are bonds or units that are not present in the pure networks, as they would indicate that more than a simple IPN mix of two networks is obtained.

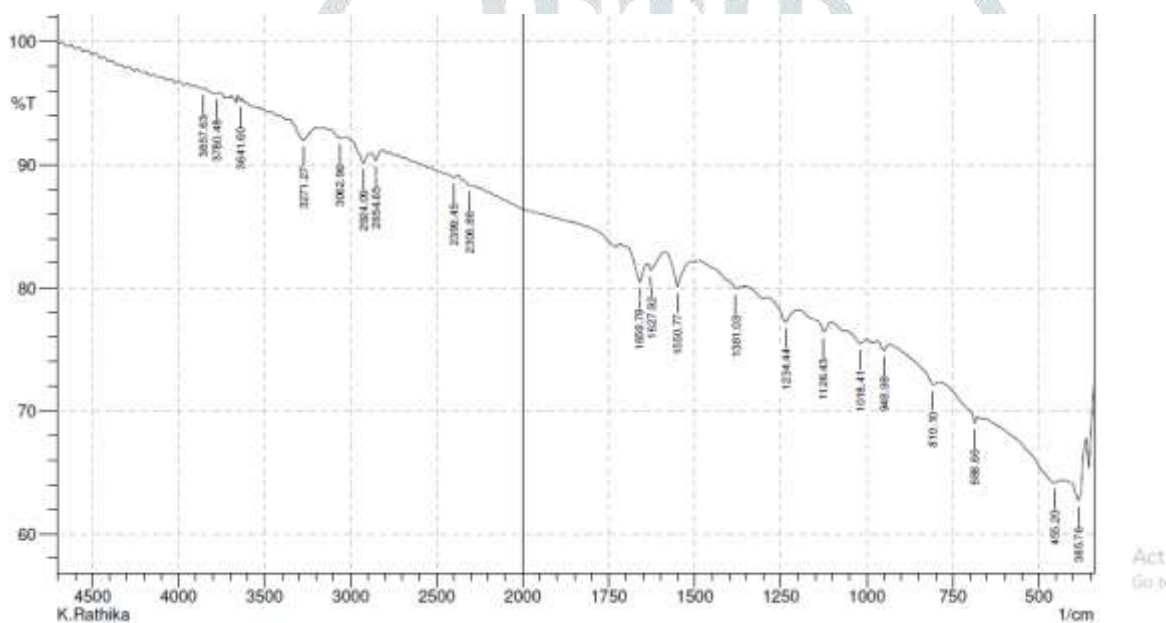
The presence of component materials in the macromolecules was confirmed by the study of FT-IR spectra. The characteristic absorptions of IPN-3CA corresponding to N-H stretching of >NH group at 3387 cm⁻¹, C-H stretchings (ss/as) of >CH₂ and -CH₃ groups at 2792.97 cm⁻¹ and 2939.52 cm⁻¹, there is no absorption band at 2260 cm⁻¹ due to NCO group of polyurethane, but C=O stretching of urethane linkage at bending at 1226.73 cm⁻¹ was observed (Prashantha et al., 2001) relied on these two confirmation signals during preparation of GC-PU/PHEMA interpenetrating polymer networks were obtained by transfer moulding²¹. The interpretations in present study are in accordance with these reported observations.

The characteristic absorptions of IPN-6CA corresponding to N-H stretching of >NH group at 3271.27 cm⁻¹, C-H stretchings (ss/as) of >CH₂ and -CH₃ groups at 2854.65 cm⁻¹ and 2924.09 cm⁻¹, there is no absorption band at 2260 cm⁻¹ due to NCO group of polyurethane, C=O stretching of urethane linkage at 1658.78 cm⁻¹. C-O bending at 1234.44.

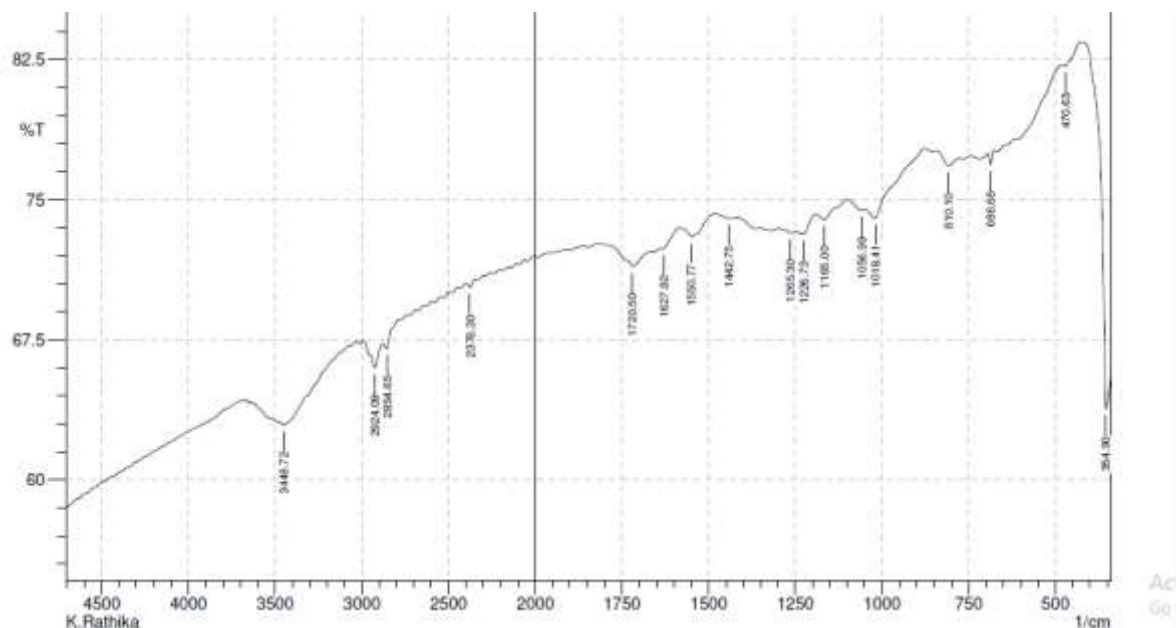
The characteristic absorptions of IPN-9CAM corresponding to >N-H stretching of =NH group at 3448.72 cm⁻¹, C-H stretchings (ss/as) of >CH₂ and -CH₃ groups at 2854.65 cm⁻¹ and 2924.09 cm⁻¹, there is no absorption band at 2260 cm⁻¹ due to NCO group of polyurethane prepolymer C=O stretching of urethane linkage at 1720.50, C-O bending at 1226.73. These are presented in Figure 1.



IPN-3CA



IPN-6CA



IPN-9CA
 Figure 1 IR- Spectrum of IPNs

Mechanical properties

The mechanical properties of the systems were evaluated using tensile testing. This provides information such as initial Young’s modulus, elongation at break, ultimate stress and strain. The mechanical properties namely tensile strength, Young's modulus, elongation at the breaking point, have been presented in Table-2.

These IPNs exhibit good mechanical properties as compared to their individual component networks I and II. Component network I is a swollen crosslinked polyurethane which is elastomeric in nature and has been used in liquid phase in this investigation. Interpenetration of this liquid polyurethane with AN monomer results in hard elastomeric films. Increase in polyurethane content in the IPN increases the extent of crosslinking in IPN and dominates the behaviour of the overall IPNs. Incorporation of polyurethane as a separate phase in the poly(AN) matrix of IPNs improved the mechanical properties. Thus it is observed that IPNs obtained from elastomeric PU and AN have quite different properties from their individual monomers. This supports further the observations made by earlier workers [23].

Table 2 Mechanical properties of IPNs

Samples	Tensile Strength(Mpa)	Elongation at break(%)	Young’s Modulus(Mpa)
IPN-1CA	2.86	148	2.54
IPN-2 CA	3.42	137	2.68
IPN-3CA	3.96	126	3.02
IPN-4CA	7.46	119	3.43

IPN-5CA	6.92	108	4.15
IPN-6CA	8.74	102	4.85
IPN-7CA	9.14	96	5.32
IPN-8CA	10.69	93	6.52
IPN-9CA	11.26	88	7.27

Chemical resistance

Chemical resistance test was carried out according to ASTM-D543-67. Degradation of biopolymers in 25% CH₃COOH, 25% H₂SO₄, 25% HCl, 5% HNO₃ and NaOH was studied and the weight loss % was estimated. The weight loss % in chemical resistance test was presented in Table 3. The weight losses of all samples were faster at pH greater than 10. It is because the base promotes hydrolysis by providing the strong nucleophilic reagent OH⁻, the degradation was faster with increase in pH. The degradation of crosslinked biopolymers was very high in sodium hydroxide. Here, the biopolymers with ester linkages can be easily hydrolyzed in the presence of alkalis.

Table 3 Weight loss % in chemical resistance test

Samples	25%CH ₃ COOH	25%H ₂ SO ₄	25%HCl	5%HNO ₃	NaOH
IPN-1CA	3.3204	1.1402	.6901	.5402	.5410
IPN-2CA	3.0253	1.2430	.6651	.6515	.6076
IPN-3CA	3.3140	1.2501	.6764	.5064	.5205
IPN-4CA	3.0205	1.1800	.6430	.5576	.5302
IPN-5CA	3.2431	1.3786	.6322	.4806	.5541
IPN-6CA	3.5424	1.8103	.6618	.4682	.5652
IPN-7CA	3.8060	1.1505	.6734	.6560	.5680
IPN-8CA	3.3240	1.6420	.7910	.6014	.5907
IPN-9CA	3.6215	1.1806	.8034	.6210	.6415

The percentage weight loss of IPNs were determined in CH₃COOH, H₂SO₄, HCl, HNO₃, and NaOH etc. the results are furnished in table 3. All the IPNs show excellent acid and alkali resistance. Nayak et al., 199724 all the IPNs show excellent acid and alkali resistance as compared to unmodified castor oil polyurethane/PHEMA IPNs. It is observed that IPNs irrespective of NCO/OH ratio and PU/PHEMA composition, are stable in acids and alkali. Thus the polymers are stable towards acids, and alkali. But in alkaline medium high weight loss % was observed. After the chemical ageing the polymer film become irregular with large number of pits, granular formation and cracks on the surface. The number of granular formation, cracks and pits becomes deeper with increasing time. IPNs like IPN-1CA, IPN-4CA and IPN-5CA shows very good chemical resistance than the other IPNs.

Stability in organic solvents

The commonly used organic solvents such as acetone, toluene, methyl ethyl ketone and dimethyl acetamide (DMA) are used in the present study. Weight loss percentages of samples were shown in Table 4. Lowest degradation rates were observed. This is due to the hydrophobic nature of this polymer network appeared by the long aliphatic hydrocarbon chain of fatty acid part of the triglyceride reported elsewhere [25]. IPNs are swelled in all the solvents like acetone, toluene, methyl ethyl ketone and dimethyl acetamide (DMA). Some weight loss and dimensional changes were observed in soft segment of polymers. These IPNs lose their gloss in acetone, toluene and methyl ethyl ketone. Among all the solvents used, the solubility parameter of dimethyl acetamide (δ_s) was found to be the solubility parameter of polyurethanes, as there was maximum swelling only in this solvent. Due to higher crosslink density the percentage of weight loss was lesser in samples like IPN-1CA and IPN-3CA. Samples like IPN-4CA and IPN-5CA show higher weight loss percentage.

Table 4 Weight loss percentage in organic solvents

Samples	NCO/OH Molar ratio	PU/PMMA weight ratio	acetone	Toluene	Methyl ethyl ketone	Dimethyl acetamide
IPN-1CA	1.6	25/75	.05	.00	.02	.14
IPN-2CA	1.6	35/65	.04	.02	.04	.12
IPN-3CA	1.6	45/55	.02	.01	.02	.08
IPN-4CA	1.8	25/75	.05	.05	.05	.10
IPN-5CA	1.8	35/65	.03	.05	.07	.12
IPN-6CA	1.8	45/55	.03	.04	.04	.08
IPN-7CA	2	25/75	.04	.03	.03	.08
IPN-8CA	2	35/65	.04	.04	.01	.06
IPN-9CA	2	45/55	.06	.05	.01	.07

THERMAL STUDIES

TGA Studies

The thermal stabilities of IPNs were studied from their thermograms, which are shown in Figure 4. For the same purpose we have chosen in the random manner, IPN-3CA, IPN-6CA and IPN-9CA.

The data showing the percentage weight loss at various temperatures are furnished in table. The thermogravimetric analysis reveals that the IPNs are thermally stable up to 250 °C with minor weight losses. These weight losses are due to loss of moisture and elimination of solvent molecules embedded in the samples. The significant weight losses of these three IPNs are observed in the temperature range of 300 °C- 500 °C, which can be attributed to the decomposition of benzene ring and decomposition of crosslinking with N-N methylene bisacrylamide between PU and MMA monomer units. The minor weight losses between 500 °C- 700 °C are due to complete decomposition of crosslinking's with N-N methylene bisacrylamide

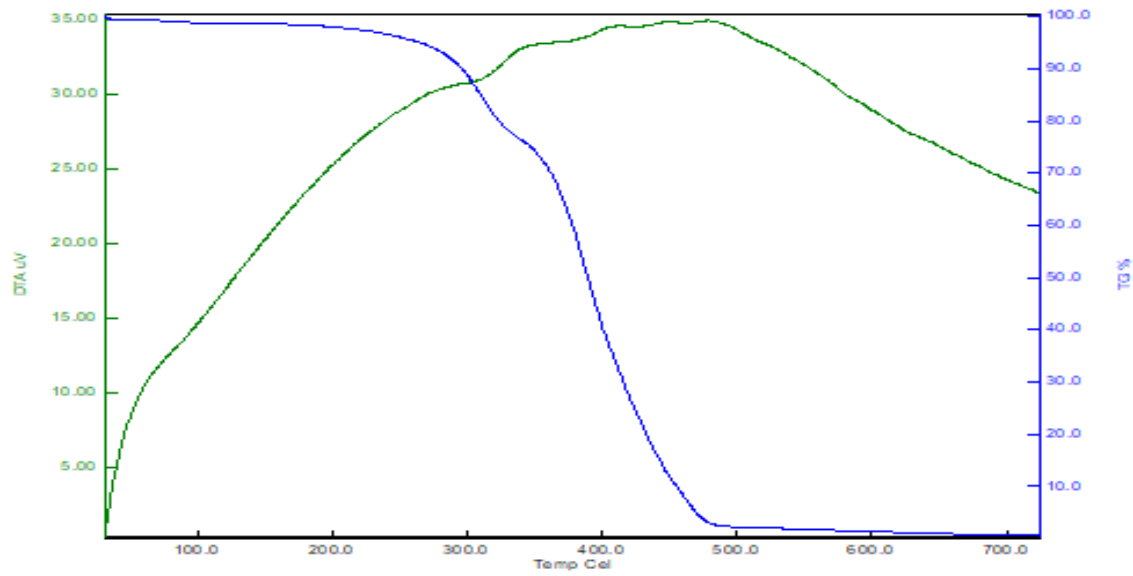
between PU and AN monomers units. Mohanty et al. 2012, which also supports the above discussed results²⁶.

Table 5 TGA data of IPNs

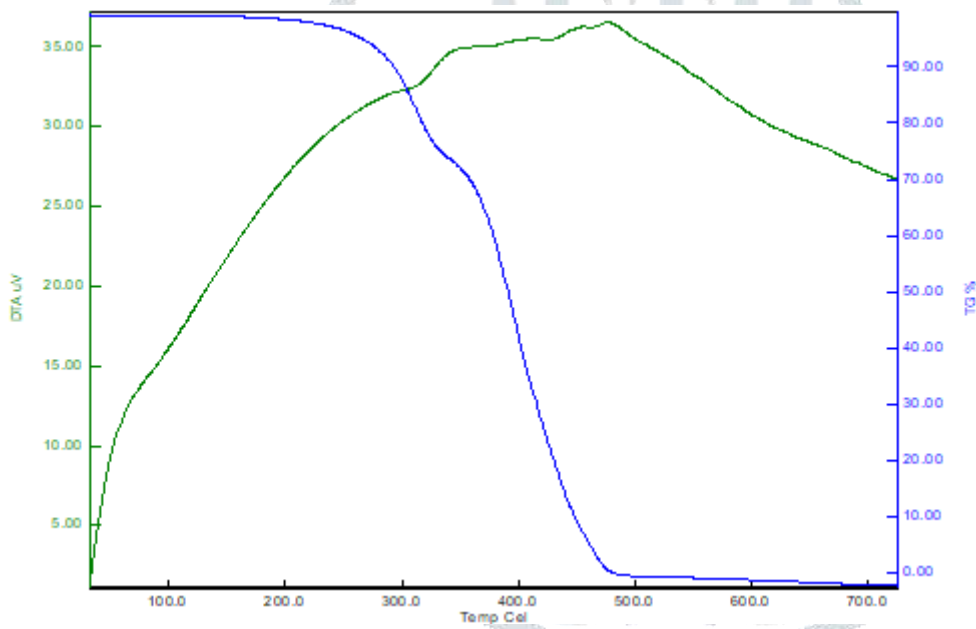
Sample code	200	300	400	500	600	700
IPN-3CA	0	20	60	35	95	99
IPN-6CA	5	30	65	30	98	-
IPN-9CA	5	20	60	31	90	99

Differential Thermal Analysis

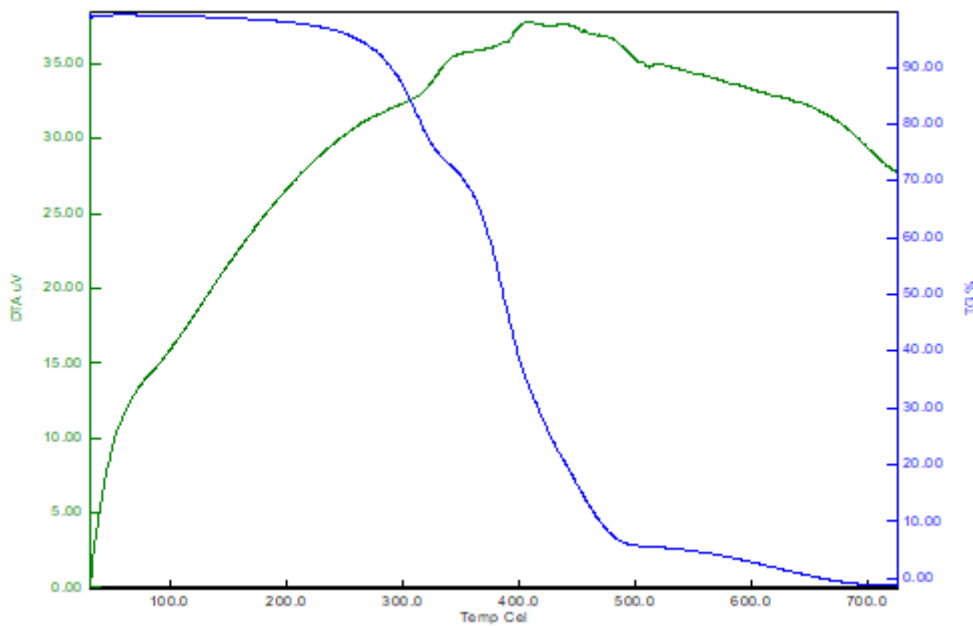
The Differential thermal analysis (DTA) of IPN samples and castor oil polyurethane are presented in Fig. 2. DTA curves show a series of degradation processes occurring in a particular temperature range. The DTA curves of IPN do not show any endothermic peak for softening. But two or three exotherms are invariably seen in the cured samples. The Differential thermal analysis data is presented in Table 5.8. In case of IPNs with NCO/OH ratio 1.6 the first exotherm is observed in the range of 330°- 360°C; second exotherm appeared nearly 420-440°C and third exotherm appeared between nearly 480-490°C. IPNs with NCO/OH ratio 1.8 show the first exotherm between 340° -360°C; second exotherm between 440° - 460°C and third exotherm is around 495°C. In the case of IPNs with NCO/OH ratio 2 the first exotherm is observed between 350° - 360°C; second exotherm appeared between 440°- 460°C and third exotherm appeared around 510°C. It has been reported^{27,28} that the exotherm around 350° C is relatively weak and is due to the cleavage of meta substituted long alkenyl side chain of the phenyl ring. The second exotherm around 440° to 480° C, which may be due to the breaking up of -CO-NH bond. The exotherm around 520° to 580° C is due to disintegration of polyurethane moiety. IPN-3CA with NCO/OH ratio 1.6 contains 45 % castor oil polyurethane shows three exotherms at 337°C, 420°C and 485°C respectively. IPN-6CA with NCO/OH ratio 1.8 contains 45 % castor oil polyurethane shows three exotherms at 359°C, 445°C and 495°C respectively. IPN-9CA with NCO/OH ratio 2.0 having 45 % castor oil polyurethane shows three exotherms at 360°C, 450°C and 510°C respectively. The thermal stability of IPNs increases with the increase of polyurethane content. Similar trend is observed in all the other IPNs.



TGA/DTA curve of IPN-3CA



TGA/DTA curve of IPN-6CA



TGA/DTA curve of IPN-9CA

Figure 2 TG/ DTA of IPNs

Table 6 DTA data of IPN polymer samples

Sample code	Composition	NCO/OH molar ratio	PU/PMMA wt. ratio	Temp.C	DTA Peak/C Exo
IPN-3CA	PU+AN	1.6	45:55	300-400 400-500	337 420 485
IPN-6CA	PU+AN	1.8	45:55	300-400 400-500	359 440 495
IPN-9CA	PU+AN	2.0	45:55	300-400 400-500	360 450 510

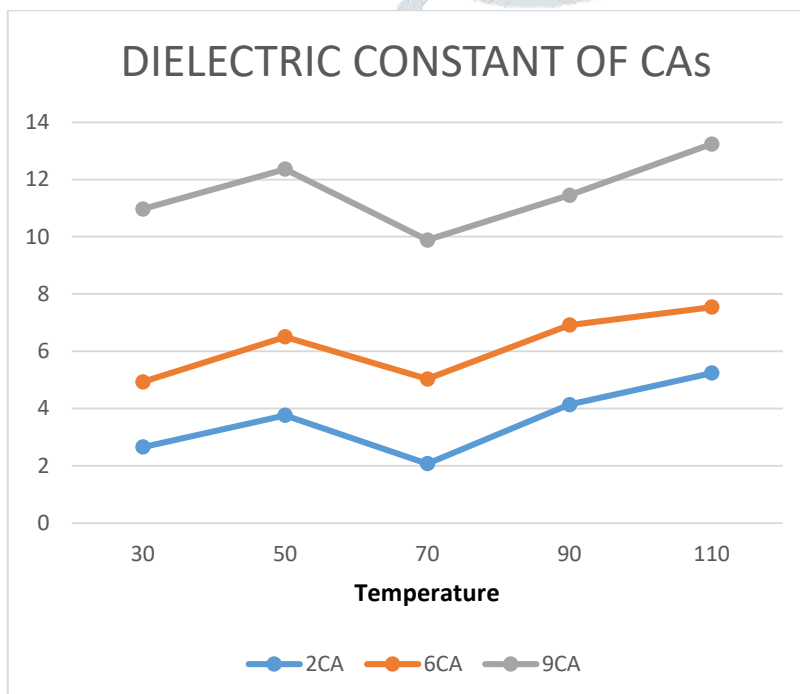
Electrical properties

Updated survey of the literature reveals that there are relatively few reports available regarding the dielectric behaviour of IPNs. The dielectric properties (ϵ' , ϵ'' , and $\tan \delta$) versus temperature at 10 KHZ frequency for IPN-3CA, IPN-6CA and IPN-9CA are shown in Figure-3. The fact that dielectric loss (ϵ'') does not change with temperature implies that IPN does not undergo attenuation when placed in dielectric field. Dielectric constant is the ability of a dielectric material to store electric potential energy under the

influence of an electric field. Dielectric constant (ϵ') is a property of insulation, which determines the electrostatic energy stored within the solid material. For all the IPNs, at a given test frequency, dielectric constant increases with increase in temperature due to softening of IPNs. At a high temperature, dielectric constant reaches a plateau and then gradually decreases (dip) at some temperature range around 70°C and later on, it increases. The gradual decrease comes from a decrease in the orientation of the dipoles as the kinetic motion of the molecules opposes the orienting effect of the dielectric field, and from the decrease in density. Loss tangent ($\tan \delta$) would also exhibit similar behaviour as it is related to dielectric constant,

$$\epsilon'' = \epsilon' \times \tan \delta$$

From these dielectrical properties, it may be concluded that the IPNs prepared behave like semiconductor and have bordering properties of insulators²⁹.



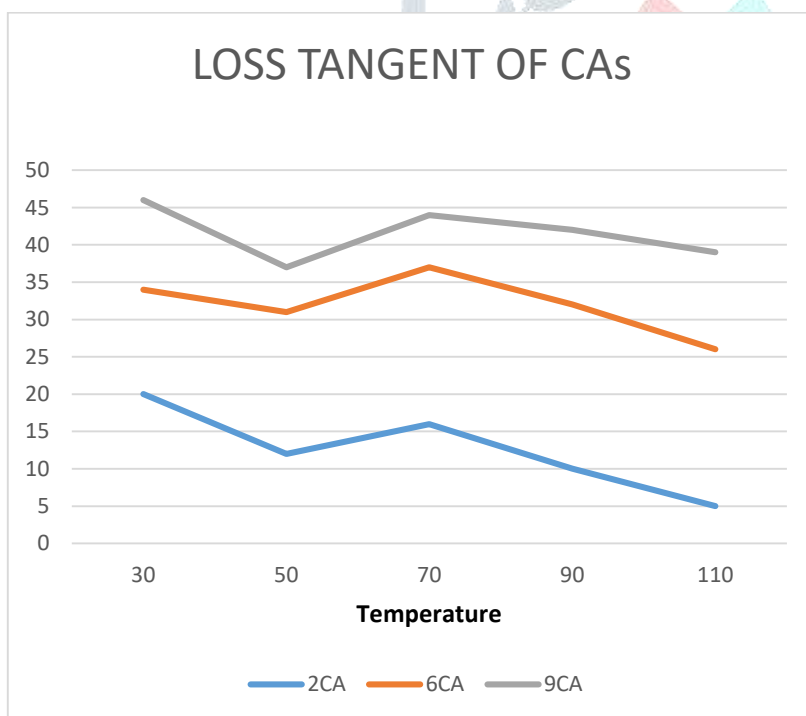
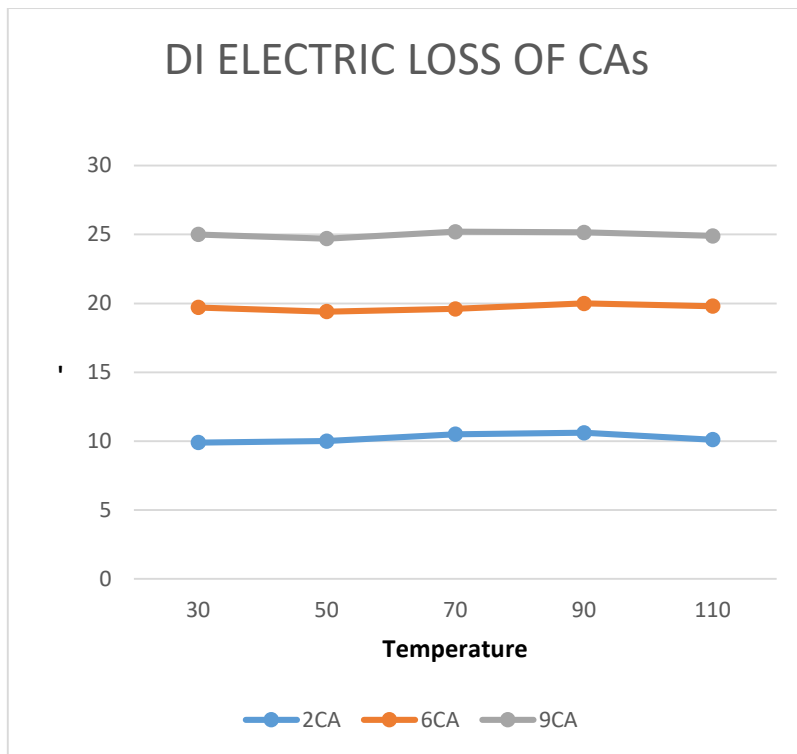


Figure 3 a) Dielectric constant of CAs b) Dielectric loss of CAs c) Loss tangent of CAs

Bio-degradation of polymers

Soil burial test

One of the serious problems, mankind is facing now, is the production and enormous use of the various types of man made polymers, which is a growing threat for the environment due to unabated

dumping. Since the polymeric materials do not decomposed easily, that is why the environmental pollution and xenobiotic nature of these polymers is a worldwide problem³⁰. Polyurethane is widely used in various fields, such as manufacture of plastic foam, cushions, rubber goods, synthetic leathers, adhesive, paints, fibers etc.

The present study deals with the isolation of polyurethane degrading microorganisms from the soil, analysis of biodegradation and optimization of various factors (temperature, pH and by using different co-metabolites). PU films were buried in soil for a period of 3 months and then used for the isolation of PU degrading microorganisms. Polyurethanes are considered to be comparatively susceptible to microbial degradation³¹. In our study, bacterial strains having the ability to utilize PU as a sole carbon source, were isolated after soil burial through enrichment in liquid medium. Kay et al. (1991) isolated 15 kinds of bacteria from polyester PU pieces following their burial in soil for 28 days³². Shah et al (2008) also isolated 12 bacterial strains, 5 through enrichment technique and 7 from buried PU films after six months³³. Soil organic carbon content and pH influence the structure of soil microorganisms³⁴. Attachment of microorganisms to buried PU is mediated by non-specific hydrophobic interaction³⁵ and local environmental conditions also influence the surface hydrophobicity of fungi and bacterial³⁶. Therefore, the differences in the physiochemical properties of the soil may influence microorganisms which successfully colonize the surface of polyurethane. Biological degradation of polymers is generally influenced by a number of factors. Besides the nature of the polymer substances, the kind of organism involved in biodegradation and environmental condition (e.g. nutrient supply, temperature and pH) are known to drastically influence the degradation rate³⁷.

Weight loss in soil burial degradation test is given in Table 5.7. The calculation of biodegradation rate from weight loss of polymer films in soil burial test constitutes a practical problem, since the soil sticks on to the film surface, and weight measurement are not accurate. Kimura et al. in their study of degradation of plastics in the soil emphasized that the degradation of plastics were mainly caused by bacteria and fungi and that different soil conditions affected the rate of degradation of plastics³⁸.

Table 7 Weight loss in soil burial degradation test

Weight loss in grams			
Samples	Initial	after 30 days	after 60 days
IPN-1CA	2.062	2.057	1.956
IPN-2CA	2.820	2.078	2.002
IPN-3CA	2.048	2.044	1.862
IPN-4CA	2.048	2.046	1.844
IPN-5CA	2.069	2.065	1.976
IPN-6CA	2.052	2.052	1.974
IPN-7CA	2.060	2.058	1.930
IPN-8CA	2.117	2.115	2.017
IPN-9CA	2.058	2.058	1.953

The loss of tensile strength, discoloration and cracking observed for soil buried in polyester polyurethane is a typical of the effects of degradation of polyester polyurethane as a result of soil burial³⁹. In our study, on the surface of the polyurethane film, there were discoloration, spots and cracking. Different patterns of degradation observed by fungi with different samples of polyurethane, were attributed to the many properties of polyurethanes such as topology and chemical composition⁴⁰. In addition, synthetic polymers generally have short repeating units. The regularity in synthetic polymers allows the polymer chains to pack easily, resulting in the formation of crystalline regions. This limits the accessibility of the polymer chains to enzymes. Huang and Roby (1986)⁴¹ observed polyurethane degradation proceeding in a selective manner, with the amorphous regions being degraded prior to the crystalline regions. It was also observed that polyurethanes with long repeating units and hydrophilic groups would be less likely to pack into high crystalline regions as normal polyurethanes and these polymers were more accessible to biodegradation. In soil degradation studies initially no weight loss was found within 20 days. After 40 days nearly 20% weight loss was occurred in all the samples. Soil degradation reached nearly 60 days weight loss of all the samples were increased to 40%. Mittal et al.,⁴² which also supports the above discussed results.

Sem morphology

Sem morphology plays a major role in affecting IPN properties, it has been extensively studied by electron microscopy and glass transition behaviour. The morphology of the IPNs depends on the method of synthesis, compatibility of polymer systems employed and on the relative rates of formation of each network. Sperling and coworkers⁴³ have reported that the morphology of simultaneous interpenetrating networks (SINs) is found to IPN the mold. Comprising the electron micrograph of IPN where the NCO/OH ratio was increased from 1.6 to 2.0 and the prepolymer-monomer ratio varies from 25:75 to 45:65, it was observed that when the monomer concentration becomes more, the AN phase becomes a continuous phase. It can be stated that the continuous dispersed phase morphology occurs only when the AN content exceeds or equal to 65%. when the PU content is less than 45%, the IPN morphology becomes complicated since both phases assume local or total continuity.

The morphology of the IPNs was studied from their SEM micrographs, which are as shown in Figure . For the same purpose we have chosen in the random manner, IPN-3CA, IPN-6CA and IPN - 9CA. Scanning electron micrographs of the IPNs show that there are two distinct phase domains due to phase separation of the two component networks⁴⁴ of IPNs. The PU network is formed first and interpenetrated by the polymethylmethacrylate component during IPN synthesis. PU is a continuous phase in which the second component is entangled. The IPNs system is phase separated, and there is dual phase continuity in the midrange composition. But due to their interlocking phase configuration, the extent of phase separation in IPNs may be restricted and the two components are intimately interpenetrating. The glass transition

temperatures also confirm the IPN formation. Sperling et al.,1983 which also supports the above discussed results.

Its sem results gave us the clear picture of cracked and biodegraded PU film. The small hole, erosion, loss of tensile strength and the roughness of the surface were clearly observed across the chemical changes such as bond formation or cleavage. Sehroon Khan et al⁴⁵ reported the same results on their work.

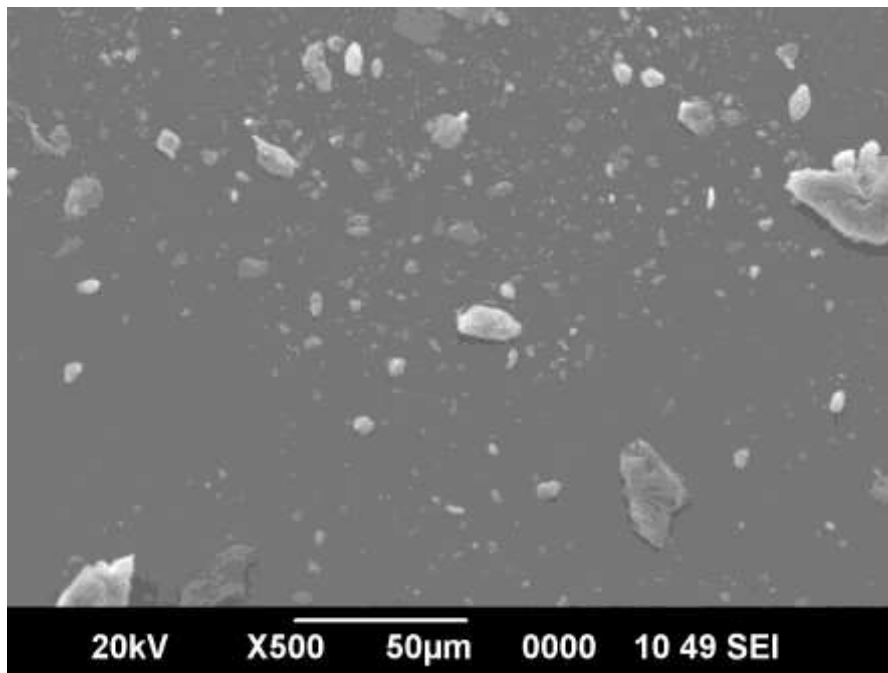


Figure 3 Sem micrograph IPN-2CA before soil burial

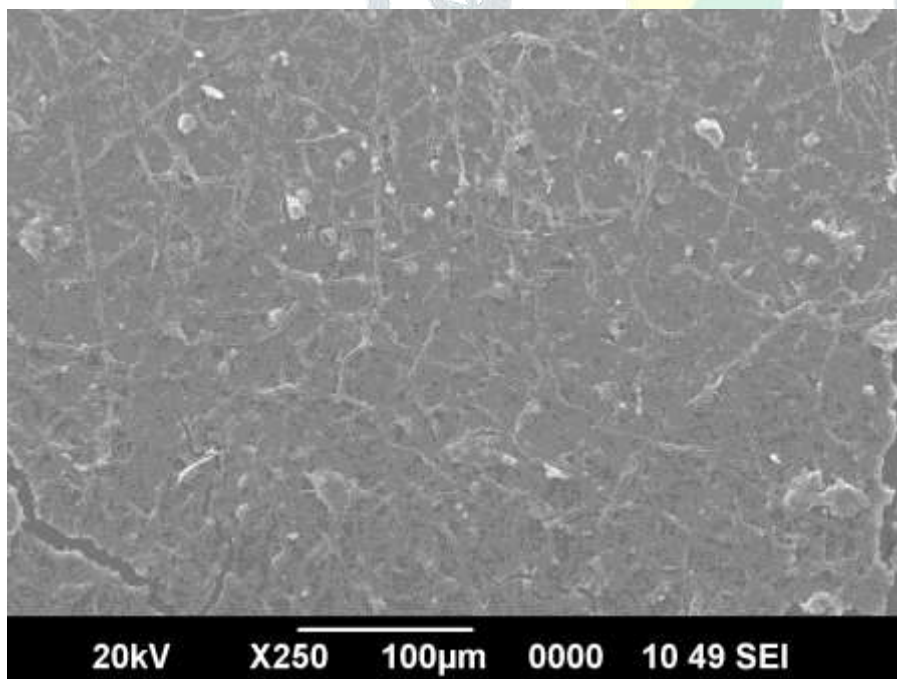


Figure 4 Sem micrograph of IPN-2CA after soil burial

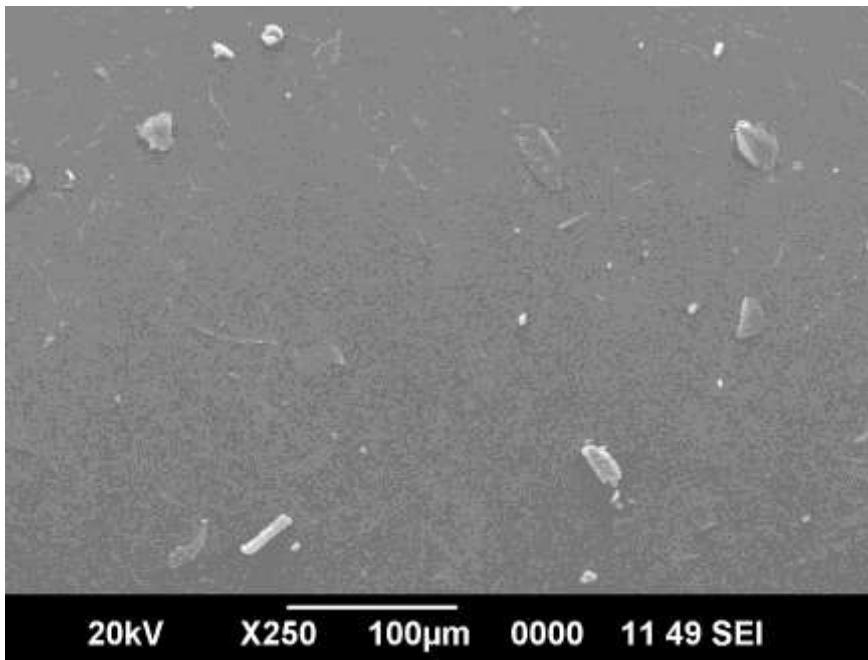


Figure 5 Sem micrograph IPN-6CA before soil burial

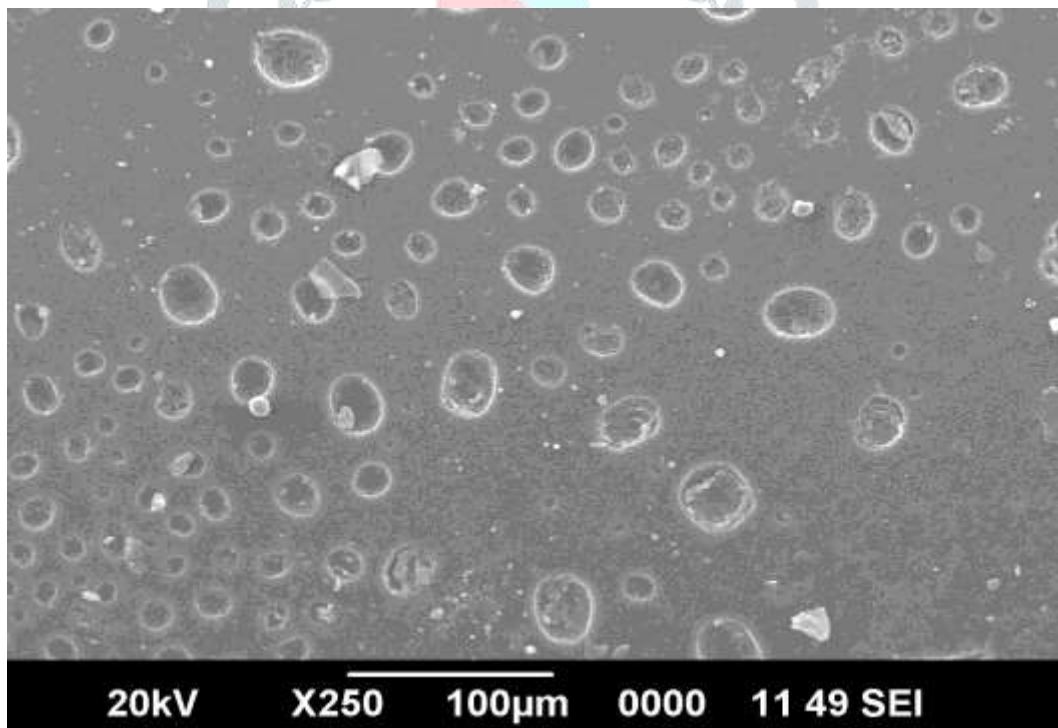


Figure 6 Sem micrograph IPN-6CA after soil burial

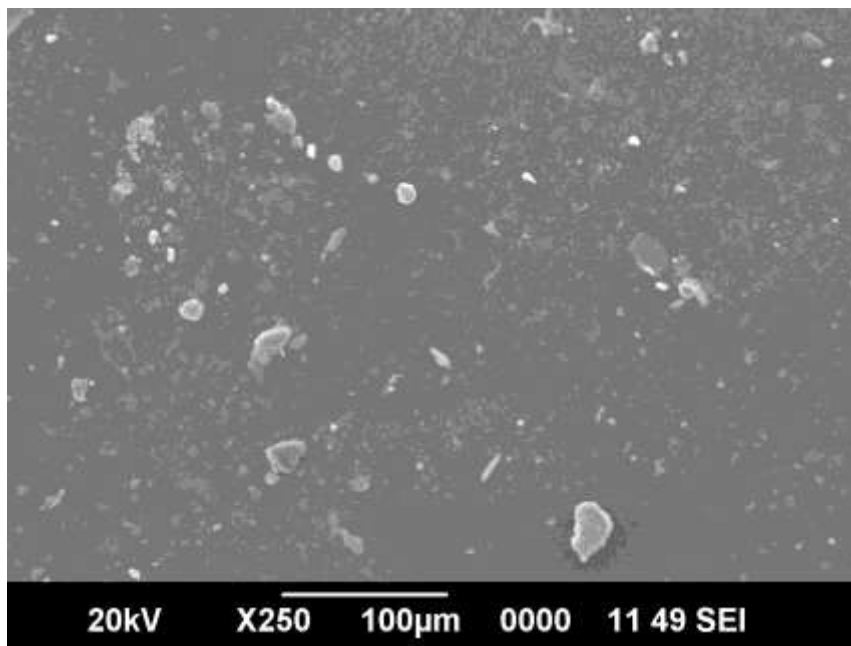


Figure 7 Sem micrograph IPN-9CAB

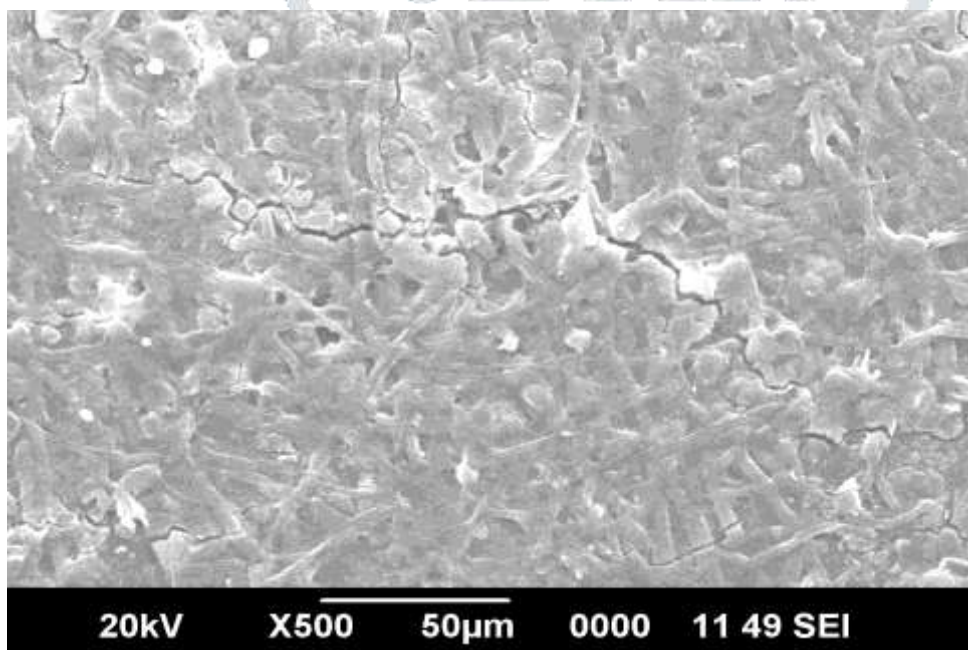


Figure 8 Sem micrograph IPN-9CAafter soil burial

CONCLUSION

These polymers have many advantages compared with polymers prepared from petroleum based monomers and in many cases are cheaper than petroleum polymers. Interpenetrating polymer networks showed excellent chemical resistance, elongation, tensile design the most desirable material for a specific end use requirement of the interpenetrating polymer networks. The major conclusion of this work is the choice of diisocyanate has a profound effect on the thermal stability of the polymer blend. Creation of IPN materials from fully polymerized immiscible polymers through a bond interchange copolymerization process, followed

bycrosslinking is a concept applicable to many other systems and opens the possibility of creating novel IPN compositions and microstructures.

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