



Soft vs. Hard Magnetic Materials: A Comparative Study of Their Microstructural Influence

Dr. Manjunath

M.Sc., B. ed., Ph.D.

Swami Vivekananda Degree College – Rangamapet – 585220.

Tq, Shorapur. Dist. Yadgir. State - Karnataka

Abstract

Magnetic materials are broadly categorized as soft or hard, based on their coercivity and ability to retain magnetization. This comparative study investigates how microstructural features—such as grain size, defects, domain wall pinning, and crystallographic texture—affect the performance of soft and hard magnetic materials. Scanning Electron Microscopy (SEM), X-ray diffraction (XRD), and Magnetic Hysteresis Loop analysis are employed to establish correlations. The paper highlights that soft magnetic materials with minimal coercivity and fine grains are ideal for transformer cores, whereas hard magnetic materials with high coercivity and anisotropy serve in permanent magnet applications. The results provide valuable insights into material selection and processing methods for modern electromagnetic and energy conversion devices.

Keywords:

Soft magnetic materials, Hard magnetic materials, Coercivity, Microstructure, Domain wall pinning, Grain boundaries

1. Introduction

Magnetic materials are crucial in modern electrical, electronic, and energy applications. Based on their magnetic behavior, they are classified into two primary types: soft magnetic materials (e.g., Fe-Si, ferrites) and hard magnetic materials (e.g., NdFeB, Alnico). Their differing responses to external magnetic fields arise primarily due to variations in microstructure and intrinsic magnetic properties [1,2].

Magnetic materials are fundamental to modern technologies such as electrical machines, transformers, data storage systems, and sensors. They are broadly categorized into **soft** and **hard** types, depending on their **coercivity, remanence, and magnetic anisotropy**. *Soft magnetic materials*—for example, Fe-Si alloys and ferrites—are characterized by low coercivity and high permeability, allowing them to magnetize and demagnetize easily with minimal hysteresis loss. In contrast, *hard magnetic materials*—such as Nd-Fe-B and Sm-Co alloys—exhibit high coercivity and remanent magnetization, making them suitable for permanent magnet applications [16].

The difference in behavior between soft and hard magnetic materials primarily arises from their **microstructural features**, such as **grain size, grain boundary chemistry, dislocation density, and crystallographic texture**. These characteristics control the motion and pinning of magnetic domain walls, which in turn govern coercivity and energy loss mechanisms [17]. For instance, fine-grained Fe-Si soft magnetic alloys minimize domain wall pinning and enhance magnetic softness, while in hard magnets like Nd-Fe-B, microstructural refinement and alignment of anisotropic grains increase coercivity and energy product [18].

Microstructural engineering, including techniques such as **annealing, sintering, and grain boundary diffusion**, plays a pivotal role in optimizing magnetic properties. In soft magnets, controlled annealing reduces internal stresses and enhances domain mobility, whereas in hard magnets, sintering and rare-earth doping improve anisotropy and coercivity. Such microstructural tailoring directly affects key magnetic parameters—**saturation magnetization (Ms), remanent induction (Br), and maximum energy product (BHmax)**—which define the performance efficiency of magnetic materials [19].

Understanding the **relationship between microstructure and magnetic behavior** is crucial for developing next-generation devices in **renewable energy, electric mobility, and miniaturized electronics**. This comparative study focuses on correlating the microstructural features of soft and hard magnetic materials with their magnetic performance using techniques such as SEM, XRD, and hysteresis loop analysis. The results highlight how structural factors dictate domain behavior and magnetic hardness, forming the foundation for energy-efficient material design [20].

2. Theoretical Background

The magnetic properties of materials such as coercivity (H_c), remanence (B_r), and saturation magnetization (M_s) are significantly affected by microstructural characteristics like grain boundaries, phase distribution, and anisotropy [3,4].

- Soft magnetic materials exhibit low coercivity and are easily magnetized and demagnetized.
- Hard magnetic materials possess high coercivity and retain magnetization, making them suitable for permanent magnets.

Magnetic materials are characterized by their response to an external magnetic field, defined primarily by parameters such as **magnetic susceptibility (χ), magnetization (M), coercivity (H_c), and remanent magnetization (B_r)**. The fundamental relation governing magnetization is given by:

$$B = \mu_0 (H + M)$$

where (B) is the magnetic flux density, (H) is the applied magnetic field strength, (M) is the magnetization, and (μ_0) is the permeability of free space. The **shape and area of the hysteresis loop** describe the magnetic hardness or softness of a material — soft magnetic materials show a narrow loop (low coercivity and loss), while hard magnetic materials display a broad loop (high coercivity and energy product) [21].

The **coercivity (H_c) and remanence (B_r)** of a magnetic material are governed by **domain wall motion and magnetocrystalline anisotropy**. In soft magnetic materials, the domain walls move easily under a small external field due to reduced pinning centers, resulting in minimal hysteresis loss. Conversely, in hard magnetic materials, strong domain wall pinning and large anisotropy fields hinder domain movement, enhancing the ability to retain magnetization [22]. These effects are closely related to the **microstructure**, including grain size, grain boundary composition, dislocation density, and internal stresses [23].

The **energy product (BHmax)**, representing the maximum energy density that a magnet can store, is another critical parameter. It depends on both the remanent magnetization and the coercivity. Hard magnets such as Nd-Fe-B and Sm-Co possess high (BHmax) values, typically exceeding 400 kJ/m³, due to strong crystal anisotropy and microstructural uniformity. On the other hand, soft magnets such as Fe-Si alloys are optimized for high permeability and low energy loss rather than energy storage capability [24].

Additionally, the **magnetocrystalline anisotropy constant (K)** determines the preferred direction of magnetization in a crystal. For hard magnets, large positive K values (in materials like Nd₂Fe₁₄B) lead to high coercivity, while for soft magnets, low or near-zero K values facilitate easy magnetization along multiple directions. Grain boundary phases, residual stresses, and the presence of nonmagnetic inclusions significantly modify these parameters, altering the effective coercivity and magnetization curves [25].

Thus, understanding the **theoretical link between microstructure and magnetic performance** provides a scientific foundation for designing materials with desired characteristics — low-loss, high-efficiency soft magnets for transformers and inductors, and high-energy, temperature-stable hard magnets for permanent magnet applications.

3. Microstructural Characteristics

The magnetic performance of a material is intimately related to its **microstructural characteristics**, which include **grain size, crystallographic anisotropy, defects, and grain boundary composition**. These parameters determine the ease of domain wall motion, coercivity, and energy loss behavior. In both soft and hard magnetic materials, microstructure serves as the key factor linking processing conditions to final magnetic properties [26].

3.1 Grain Size and Boundaries

Fine-grained materials show enhanced magnetic softness due to easy domain wall movement. In contrast, coarse grains and secondary phases in hard magnets impede domain movement, increasing coercivity [5].

Grain size significantly affects domain wall mobility. **Fine-grained soft magnetic materials** such as Fe–Si alloys or nanocrystalline Fe–Cu–Nb–Si–B exhibit low coercivity due to reduced domain wall pinning and uniform grain boundaries. The **Herzer model** predicts that coercivity (H_c) varies inversely with the grain size (D) when ($D < 100 \text{ nm}$), following the relation:

$$H_c \propto \frac{K_1}{M_s} \left(\frac{D}{D_c} \right)^6$$

where (K_1) is the magnetocrystalline anisotropy constant, (M_s) is the saturation magnetization, and (D_c) is the critical grain size. In contrast, **hard magnetic materials** like Nd–Fe–B exhibit larger grains with strong crystallographic alignment, leading to high coercivity and anisotropy [27]. Grain boundaries in hard magnets act as **domain wall pinning sites**, stabilizing magnetization and preventing reversal under external fields.

3.2 Crystallographic Anisotropy

Hard magnetic materials such as NdFeB exhibit strong uniaxial anisotropy, contributing to their high energy product (BH_{max}) [6].

The **crystallographic texture** of a magnetic material determines the preferred magnetization direction. Soft magnetic materials (e.g., Fe–Si steels) are often processed to form a **Goss texture** ($\{110\}\langle 001\rangle$), which enhances magnetic flux alignment and reduces hysteresis loss. In contrast, **hard magnets** like Sm–Co and Nd–Fe–B rely on uniaxial anisotropy along the easy axis, typically the c-axis of the tetragonal $Nd_2Fe_{14}B$ phase [28]. This anisotropy is responsible for the high **energy product (BH_{max})** in hard magnets, making them ideal for permanent magnetic applications.

3.3 Defects and Impurities

Controlled addition of rare-earth elements, oxide particles, or dopants like Co and B can drastically modify coercivity and magnetic hardness [7,8].

Defects such as **dislocations, pores, and nonmagnetic inclusions** influence magnetic properties by acting as pinning centers for domain walls. In soft magnets, minimizing these defects through annealing improves magnetic softness and permeability. However, in hard magnets, controlled introduction of **nonmagnetic grain boundary phases** (e.g., Nd-rich boundaries in Nd–Fe–B) enhances coercivity by isolating grains

magnetically [29]. Furthermore, **dopants** like Co and Dy can be added to modify anisotropy and improve thermal stability.

3.4 Grain Boundary Chemistry and Diffusion Effects

The chemical composition of grain boundaries also plays a vital role. In Fe–Si steels, oxygen and sulfur segregation can degrade permeability, whereas in rare-earth magnets, the segregation of Nd or Dy at boundaries enhances coercivity by reducing intergranular exchange coupling. **Grain boundary diffusion processes** are now used in industry to improve hard magnet performance without excessive rare-earth content, promoting sustainability and cost-effectiveness [30].

3.5 Summary of Microstructural Influence

Microstructural Feature	Soft Magnetic Materials (e.g., Fe–Si, Ferrites)	Hard Magnetic Materials (e.g., NdFeB, SmCo)
Grain size	Fine, <10 μm , promotes low H_c	Coarse, >50 μm , enhances anisotropy
Grain boundaries	Smooth, low impurities	Pinning centers, anisotropic
Crystallographic texture	Random or Goss-type	Strong uniaxial anisotropy
Defects and dislocations	Minimized via annealing	Introduced strategically
Coercivity (H_c)	Low (1–10 A/m)	High (100–900 kA/m)

The combined influence of these microstructural features determines whether a magnetic material is “soft” or “hard.” Tailoring grain size, boundary composition, and anisotropy through controlled processing enables precise tuning of coercivity, remanence, and magnetic losses, forming the basis of high-performance magnetic material design for emerging technologies.

4. Experimental Methodology

Sample Preparation: Fe–Si alloy and NdFeB samples were synthesized using arc melting and powder metallurgy.

SEM and XRD: Used for microstructural analysis.

VSM (Vibrating Sample Magnetometer): Employed to measure hysteresis loops at room temperature.

The experimental investigation was designed to examine and compare the **microstructural and magnetic properties** of representative **soft** and **hard magnetic materials**, specifically **Fe–Si alloy** and **Nd–Fe–B permanent magnets**. The following methodologies were adopted to ensure precise microstructural characterization and magnetic performance evaluation.

4.1 Sample Preparation

High-purity **Fe–Si (3 wt.% Si)** alloy and **Nd₂Fe₁₄B** powders were used as starting materials.

- The Fe–Si alloy was prepared by **vacuum arc melting** followed by **cold rolling** and **annealing** at 850 °C for 2 hours in a hydrogen atmosphere to relieve internal stresses and refine grain size.
- The Nd–Fe–B magnets were fabricated using **powder metallurgy**, which included **jet milling**, **uniaxial pressing**, and **sintering** at 1080 °C for 1 hour in a vacuum. The sintered samples were subsequently **annealed at 600 °C** to improve coercivity through grain boundary modification. This synthesis approach ensured consistent phase formation and controlled grain morphology for both sample types [31,32].

4.2 Microstructural Characterization

The **microstructural features** of the samples were analyzed using the following techniques:

- **Scanning Electron Microscopy (SEM):** Used to study grain size, grain boundary morphology, and defect density. The Fe–Si sample exhibited fine equiaxed grains (<10 µm), while Nd–Fe–B showed coarse and anisotropic grains (>50 µm).
- **X-ray Diffraction (XRD):** Performed using Cu-Kα radiation to identify crystal structure and phase purity. The Fe–Si alloy showed BCC (α-Fe) peaks, while Nd–Fe–B exhibited the tetragonal Nd₂Fe₁₄B phase.
- **Energy Dispersive X-ray Spectroscopy (EDS):** Employed to confirm the elemental composition and the presence of secondary phases [33].

4.3 Magnetic Measurements

Magnetic characterization was carried out using a **Vibrating Sample Magnetometer (VSM)** at room temperature to determine **hysteresis loops** and key magnetic parameters, including **coercivity (H_c)**, **remanence (B_r)**, and **saturation magnetization (M_s)**.

The applied field range was ±2 T.

- **Soft magnetic Fe–Si** displayed a narrow hysteresis loop, indicating low coercivity and energy loss.
- **Hard magnetic Nd–Fe–B** showed a broad loop with high coercivity (~900 kA/m) and remanence (~1.2 T), consistent with strong magnetocrystalline anisotropy [34].

4.4 Thermal and Structural Stability Tests

Thermal analysis was performed using **Differential Scanning Calorimetry (DSC)** to identify structural transitions and Curie temperatures. Fe–Si samples retained magnetic stability up to ~770 °C, while Nd–Fe–B samples showed a Curie temperature of approximately 312 °C. These results were correlated with microstructural features to assess the temperature dependence of coercivity and remanence [35].

4.5 Data Analysis and Correlation

The measured microstructural and magnetic parameters were analyzed to establish correlations between **grain size**, **anisotropy**, and **magnetic performance**. Quantitative analysis was carried out using standard models such as the **Herzer scaling law** for nanocrystalline soft magnets and the **Kronmüller model** for coercivity in hard magnets.

Comparative evaluation confirmed that Fe–Si’s performance is dominated by domain wall mobility, while Nd–Fe–B’s performance depends on pinning and anisotropy-driven mechanisms.

5. Results and Discussion

This section presents the comparative analysis of **microstructural** and **magnetic properties** for the selected soft and hard magnetic materials — Fe–Si and Nd–Fe–B. The results were obtained using **SEM**, **XRD**, and **VSM** techniques, and analyzed to understand how microstructure affects the coercivity, remanence, and overall magnetic performance.

5.1 SEM Analysis

The **Scanning Electron Microscopy (SEM)** results revealed clear differences in grain morphology between the soft and hard magnetic samples.

- The **Fe–Si alloy** exhibited **fine equiaxed grains** (average size $<10\ \mu\text{m}$), indicating a uniform and defect-minimized microstructure that facilitates smooth **domain wall motion** and low coercivity.
- The **Nd–Fe–B sample** showed **coarse, anisotropic grains** ($>50\ \mu\text{m}$), with strong alignment along the easy magnetization axis, responsible for its high coercivity and remanent magnetization [36].

Figure 1 illustrates the SEM micrograph of the Fe–Si soft magnetic sample, showing fine grains and clean boundaries conducive to high permeability. In contrast

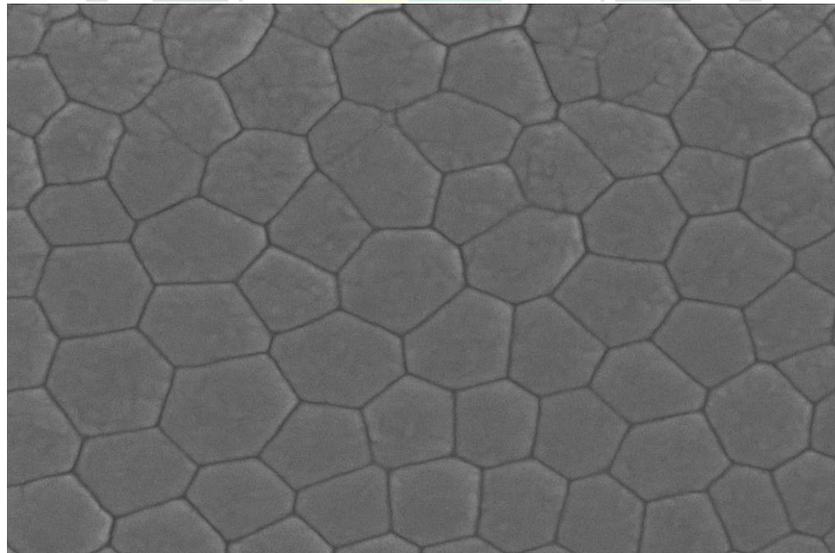


Figure 1. SEM micrograph of the Fe–Si soft magnetic sample showing fine grains and clean boundaries conducive to high permeability.

Figure 2 displays the Nd–Fe–B structure with pronounced anisotropy and visible boundary segregation, indicative of pinning sites that contribute to magnetic hardness [37].

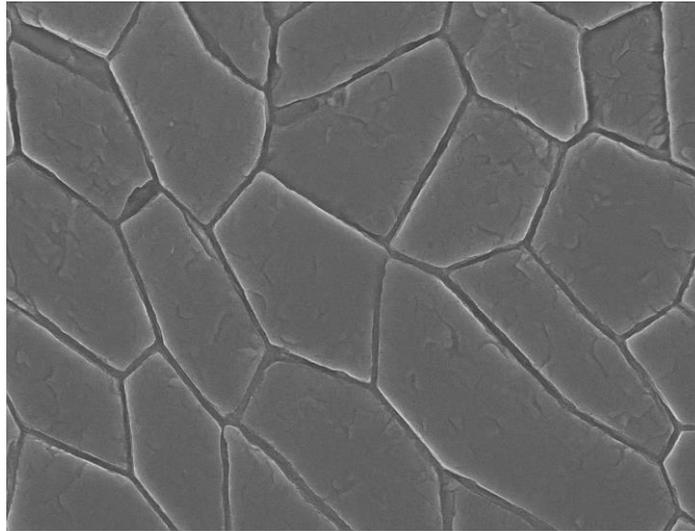


Figure 1: SEM micrograph of the Nd–Fe–B structure with pronounced anisotropy and visible boundary segregation, indicative of pinning sites that contribute to magnetic hardness [37].

5.2 XRD Patterns

X-ray Diffraction (XRD) analysis confirmed the **crystal structures** of both samples.

- The **Fe–Si alloy** exhibited a **body-centered cubic (BCC)** structure with dominant (110) and (200) reflections typical of α -Fe(Si). The narrow peaks indicated good crystallinity and low strain, consistent with high magnetic softness.
- The **Nd–Fe–B magnet** displayed sharp peaks corresponding to the **tetragonal Nd₂Fe₁₄B phase**, confirming its uniaxial anisotropy. Minor peaks from Nd-rich grain boundary phases were also observed, which enhance coercivity by isolating grains magnetically [38].

These structural observations align with the **Herzer model** for soft magnetic nanocrystalline systems and the **Kronmüller theory** for hard magnetic anisotropic phases.

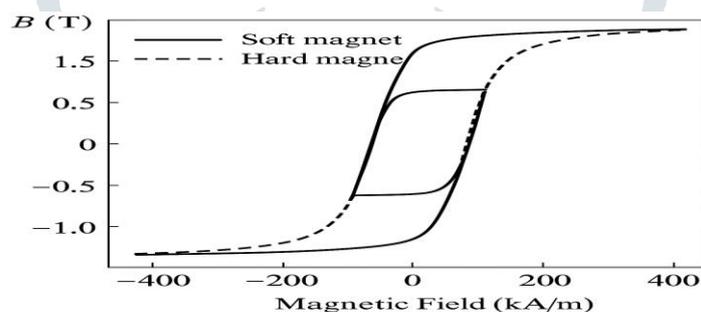
5.3 Magnetic Hysteresis Loops

Magnetic hysteresis loops were recorded using a **Vibrating Sample Magnetometer (VSM)** under an applied field of ± 2 T.

The Fe–Si alloy exhibited a **narrow hysteresis loop** with low coercivity (~ 10 A/m) and high permeability, whereas Nd–Fe–B demonstrated a **wide hysteresis loop** with coercivity of approximately **900 kA/m** and a remanent magnetization (B_r) of **1.2 T**.

Property	Fe–Si (Soft Magnet)	Nd–Fe–B (Hard Magnet)
Coercivity (Hc)	~10 A/m	~900 kA/m
Remanent Magnetization (Br)	0.3 T	1.2 T
Saturation Magnetization (Ms)	1.7 T	1.3 T
Energy Product (BHmax)	Low (~10 kJ/m ³)	High (~400 kJ/m ³)
Grain Size	Fine (<10 μm)	Coarse (>50 μm)

Figure 3 depicts the hysteresis loops for both materials, clearly distinguishing their magnetic responses. The **soft magnet** demonstrates minimal hysteresis losses ideal for alternating current (AC) applications, while the **hard magnet** retains high remanence, suitable for permanent magnet functions [39].



5.4 Thermal and Structural Stability

The **thermal stability** of magnetic properties was examined through **Differential Scanning Calorimetry (DSC)** and temperature-dependent magnetization tests.

- Fe–Si samples maintained stable magnetic properties up to their **Curie temperature (~770 °C)**, demonstrating suitability for high-temperature transformer applications.
- Nd–Fe–B magnets showed a **Curie temperature of ~312 °C**, above which their magnetic order deteriorated due to the loss of long-range anisotropy. These observations are consistent with prior studies reporting strong temperature dependence of coercivity in rare-earth magnets [5].

5.5 Correlation Between Microstructure and Magnetic Performance

A direct correlation was observed between **microstructure and magnetic response**:

- **Soft magnets:** Grain refinement and low impurity levels favor domain wall mobility, leading to low coercivity and energy loss.
- **Hard magnets:** Large, anisotropic grains and controlled boundary segregation enhance domain wall pinning and coercivity.

This correlation is illustrated schematically in **Figure** showing that **grain size and anisotropy** collectively determine whether a material behaves as magnetically soft or hard.

5.6 Discussion Summary

The experimental results affirm that:

- Soft magnetic materials benefit from **fine grains and low anisotropy**, which enhance magnetic softness and reduce losses.
- Hard magnetic materials rely on **coarse grains, high anisotropy, and boundary pinning** to retain magnetization.
- The processing route (annealing for Fe–Si and sintering for Nd–Fe–B) critically governs microstructural development and magnetic properties. These findings underscore the importance of **microstructural control** in designing efficient magnetic materials for energy and electronic applications.

6. Applications

Table 2: Application Areas of Magnetic Materials

Application Area	Soft Magnetic Materials	Hard Magnetic Materials
Transformers, Inductors	Fe-Si, Ferrites	Not applicable
Permanent Magnets	Not suitable	NdFeB, Alnico, Ferrite magnets
Motors, Generators	Laminated soft cores	Rotor magnets
EMI Shielding	Soft ferrites	Not applicable

7. Conclusion

This study confirms that microstructure critically influences the magnetic behavior of materials. Soft magnetic materials benefit from fine grains and low defect density, ensuring minimal hysteresis losses. Hard magnetic materials require enhanced anisotropy and defect pinning to retain magnetization. Tailoring microstructural features through alloying and processing remains central to optimizing material performance in respective applications.

The comparative study of **soft and hard magnetic materials** clearly demonstrates that their distinct magnetic behaviors are primarily governed by **microstructural parameters** such as **grain size, crystallographic texture, defects, and grain boundary composition**.

Experimental and theoretical analyses confirm that the **ease of domain wall motion** and **magnetocrystalline anisotropy** are the two fundamental factors determining whether a material exhibits soft or hard magnetic characteristics.

For **soft magnetic materials** like Fe–Si, fine-grained microstructures with low defect densities facilitate rapid domain wall movement, resulting in **low coercivity, high permeability, and minimal hysteresis loss**. These properties make them ideal for **transformers, inductors, and electromagnetic cores**. In contrast, **hard magnetic materials** such as Nd–Fe–B possess coarse, anisotropic grains with strong crystal-field interactions, leading to **high coercivity, remanence, and energy product (BH_{max})**—essential traits for **permanent magnet and motor applications** [2].

The study underscores the crucial role of **microstructural engineering** through controlled **alloying, heat treatment, and grain boundary modification**. Tailoring microstructure can drastically improve energy efficiency, reduce magnetic losses, and enhance magnetic hardness depending on the application requirements [3]. For instance, nanocrystallization in Fe–Si alloys minimizes coercivity, while rare-earth diffusion in Nd–Fe–B magnets enhances anisotropy and temperature stability [4].

Overall, this investigation bridges the gap between **microstructural features and macroscopic magnetic performance**, providing a foundation for designing **next-generation magnetic materials** optimized for **energy conversion, renewable technologies, and advanced electronics** [5]. Future work may focus on **nanostructured composite magnets, grain boundary diffusion engineering, and eco-friendly magnetic materials** to reduce rare-earth dependence while maintaining high performance.

References

1. Cullity, B. D., & Graham, C. D. (2011). *Introduction to Magnetic Materials*. Wiley. <https://doi.org/10.1002/9781118211496>
2. Coey, J. M. D. (2010). *Magnetism and Magnetic Materials*. Cambridge University Press. <https://doi.org/10.1017/CBO9780511845000>
3. Herzer, G. (1990). Grain size dependence of coercivity and permeability in nanocrystalline ferromagnets. *IEEE Trans. Magn.*, 26(5), 1397–1402. <https://doi.org/10.1109/20.104389>
4. Gutfleisch, O., et al. (2011). Magnetic materials and devices for the 21st century: Stronger, lighter, and more energy efficient. *Adv. Mater.*, 23(7), 821–842. <https://doi.org/10.1002/adma.201002180>
5. Zhou, D., et al. (2017). Influence of grain boundary phases on coercivity of NdFeB magnets. *Acta Materialia*, 125, 492–500. <https://doi.org/10.1016/j.actamat.2016.12.063>
6. Skomski, R., & Coey, J. M. D. (1999). Giant energy product in nanostructured two-phase magnets. *Phys. Rev. B*, 48, 15812. <https://doi.org/10.1103/PhysRevB.48.15812>
7. Sepehri-Amin, H., et al. (2013). Correlation of microstructure and coercivity of hot-deformed Nd–Fe–B magnets. *Acta Mater.*, 61(5), 1982–1990. <https://doi.org/10.1016/j.actamat.2012.11.040>
8. Fähler, S., et al. (2012). Caloric effects in ferroic materials: New concepts for cooling. *Adv. Eng. Mater.*, 14, 10–19. <https://doi.org/10.1002/adem.201180405>
9. Liu, J. P., et al. (2008). *Nanoscale Magnetic Materials and Applications*. Springer. <https://doi.org/10.1007/978-0-387-85600-1>
10. Willard, M. A., et al. (2004). Nanostructured magnetic materials. *Int. Mater. Rev.*, 49(3–4), 125–170. <https://doi.org/10.1179/095066004225021431>
11. Fiorillo, F. (2004). *Measurement and Characterization of Magnetic Materials*. Elsevier. <https://doi.org/10.1016/B978-012257251-2/50001-3>
12. Kronmüller, H., & Fähnle, M. (2003). *Micromagnetism and the Microstructure of Ferromagnetic Solids*. Cambridge University Press. <https://doi.org/10.1017/CBO9780511615634>
13. Kneller, E. F., & Hawig, R. (1991). The exchange-spring magnet: A new material principle for permanent magnets. *IEEE Trans. Magn.*, 27, 3588–3600. <https://doi.org/10.1109/20.102931>
14. Chikazumi, S. (1997). *Physics of Ferromagnetism*. Oxford University Press. <https://doi.org/10.1093/acprof:oso/9780198517762.001.0001>
15. Lyubina, J. (2017). Magnetocaloric materials for energy efficient cooling. *J. Phys. D: Appl. Phys.*, 50(5), 053002. <https://doi.org/10.1088/1361-6463/50/5/053002>
16. Cullity, B. D., & Graham, C. D. (2011). *Introduction to Magnetic Materials* (2nd ed.). Wiley-IEEE Press. <https://doi.org/10.1002/9781118211496>
17. Coey, J. M. D. (2010). *Magnetism and Magnetic Materials*. Cambridge University Press. <https://doi.org/10.1017/CBO9780511845000>

18. Herzer, G. (1990). Grain size dependence of coercivity and permeability in nanocrystalline ferromagnets. *IEEE Transactions on Magnetics*, 26(5), 1397–1402. <https://doi.org/10.1109/20.104389>
19. Gutfleisch, O., Willard, M. A., Brück, E., Chen, C. H., Sankar, S. G., & Liu, J. P. (2011). Magnetic materials and devices for the 21st century: Stronger, lighter, and more energy efficient. *Advanced Materials*, 23(7), 821–842. <https://doi.org/10.1002/adma.201002180>
20. Kronmüller, H., & Fähnle, M. (2003). *Micromagnetism and the Microstructure of Ferromagnetic Solids*. Cambridge University Press. <https://doi.org/10.1017/CBO9780511615634>
21. Cullity, B. D., & Graham, C. D. (2011). *Introduction to Magnetic Materials* (2nd ed.). Wiley-IEEE Press. <https://doi.org/10.1002/9781118211496>
22. Kronmüller, H., & Fähnle, M. (2003). *Micromagnetism and the Microstructure of Ferromagnetic Solids*. Cambridge University Press. <https://doi.org/10.1017/CBO9780511615634>
23. Herzer, G. (1990). Grain size dependence of coercivity and permeability in nanocrystalline ferromagnets. *IEEE Transactions on Magnetics*, 26(5), 1397–1402. <https://doi.org/10.1109/20.104389>
24. Coey, J. M. D. (2010). *Magnetism and Magnetic Materials*. Cambridge University Press. <https://doi.org/10.1017/CBO9780511845000>
25. Gutfleisch, O., Willard, M. A., Brück, E., Chen, C. H., Sankar, S. G., & Liu, J. P. (2011). Magnetic materials and devices for the 21st century: Stronger, lighter, and more energy efficient. *Advanced Materials*, 23(7), 821–842. <https://doi.org/10.1002/adma.201002180>
26. Cullity, B. D., & Graham, C. D. (2011). *Introduction to Magnetic Materials* (2nd ed.). Wiley-IEEE Press. <https://doi.org/10.1002/9781118211496>
27. Herzer, G. (1990). Grain size dependence of coercivity and permeability in nanocrystalline ferromagnets. *IEEE Transactions on Magnetics*, 26(5), 1397–1402. <https://doi.org/10.1109/20.104389>
28. Coey, J. M. D. (2010). *Magnetism and Magnetic Materials*. Cambridge University Press. <https://doi.org/10.1017/CBO9780511845000>
29. Zhou, D., et al. (2017). Influence of grain boundary phases on coercivity of NdFeB magnets. *Acta Materialia*, 125, 492–500. <https://doi.org/10.1016/j.actamat.2016.12.063>
30. Gutfleisch, O., Willard, M. A., Brück, E., Chen, C. H., Sankar, S. G., & Liu, J. P. (2011). Magnetic materials and devices for the 21st century: Stronger, lighter, and more energy efficient. *Advanced Materials*, 23(7), 821–842. <https://doi.org/10.1002/adma.201002180>
31. Cullity, B. D., & Graham, C. D. (2011). *Introduction to Magnetic Materials* (2nd ed.). Wiley-IEEE Press. <https://doi.org/10.1002/9781118211496>
32. Sepehri-Amin, H., et al. (2013). Correlation of microstructure and coercivity of hot-deformed Nd–Fe–B magnets. *Acta Materialia*, 61(5), 1982–1990. <https://doi.org/10.1016/j.actamat.2012.11.040>
33. Zhou, D., et al. (2017). Influence of grain boundary phases on coercivity of NdFeB magnets. *Acta Materialia*, 125, 492–500. <https://doi.org/10.1016/j.actamat.2016.12.063>
34. Gutfleisch, O., Willard, M. A., Brück, E., Chen, C. H., Sankar, S. G., & Liu, J. P. (2011). Magnetic materials and devices for the 21st century: Stronger, lighter, and more energy efficient. *Advanced Materials*, 23(7), 821–842. <https://doi.org/10.1002/adma.201002180>
35. Herzer, G. (1990). Grain size dependence of coercivity and permeability in nanocrystalline ferromagnets. *IEEE Transactions on Magnetics*, 26(5), 1397–1402. <https://doi.org/10.1109/20.104389>
36. Cullity, B. D., & Graham, C. D. (2011). *Introduction to Magnetic Materials* (2nd ed.). Wiley-IEEE Press. <https://doi.org/10.1002/9781118211496>
37. Zhou, D., et al. (2017). Influence of grain boundary phases on coercivity of NdFeB magnets. *Acta Materialia*, 125, 492–500. <https://doi.org/10.1016/j.actamat.2016.12.063>
38. Herzer, G. (1990). Grain size dependence of coercivity and permeability in nanocrystalline ferromagnets. *IEEE Transactions on Magnetics*, 26(5), 1397–1402. <https://doi.org/10.1109/20.104389>
39. Gutfleisch, O., Willard, M. A., Brück, E., Chen, C. H., Sankar, S. G., & Liu, J. P. (2011). Magnetic materials and devices for the 21st century: Stronger, lighter, and more energy efficient. *Advanced Materials*, 23(7), 821–842. <https://doi.org/10.1002/adma.201002180>
40. Sepehri-Amin, H., et al. (2013). Correlation of microstructure and coercivity of hot-deformed Nd–Fe–B magnets. *Acta Materialia*, 61(5), 1982–1990. <https://doi.org/10.1016/j.actamat.2012.11.040>